

Article

Defective TiO₂ Core-Shell Magnetic Photocatalyst Modified with Plasmonic Nanoparticles for Visible Light-Induced Photocatalytic Activity

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Abstract: In the presented work, for the first time, the metal-modified defective titanium(IV) oxide nanoparticles with well-defined titanium vacancies, was successfully obtained. Introducing platinum and copper nanoparticles (NPs) as surface modifiers of defective d-TiO₂ significantly increased the photocatalytic activity in both UV-Vis and Vis light ranges. Moreover, metal NPs deposition on the magnetic core allowed for the effective separation and reuse of the nanometer-sized photocatalyst from the suspension after the treatment process. The obtained Fe₃O₄@SiO₂/d-TiO₂-Pt/Cu photocatalysts were characterized by X-ray diffractometry (XRD) and specific surface area (BET) measurements, UV-Vis diffuse reflectance spectroscopy (DR-UV/Vis), X-ray photoelectron spectroscopy (XPS) and transmission electron microscopy (TEM). Further, the mechanism of phenol degradation and the role of four oxidative species (h^+ , e^- , \bullet OH, and \bullet O₂⁻) in the studied photocatalytic process were investigated.

Keywords: titanium vacancies; phenol degradation; scavengers; magnetic photocatalysts; platinum-modified defective TiO₂

1. Introduction

In recent years, among wastewater treatment and environmental remediation technologies, photocatalysis has gained attention as a promising technique for the degradation of persistent organic pollutants at ambient temperature and pressure [1–4]. Pilot scale-installations for water treatment using photocatalysis are more and more popular among the world [5,6]. The structural and surface properties of photocatalysts significantly influence their physicochemical and photocatalytic properties. In this regard, one of the most important issues in the photocatalytic process is the preparation of well-characterized and highly active photocatalytic material.

Titanium(IV) oxide (TiO₂), the most widely used semiconductor in photocatalysis, is extensively exploited to obtain highly photoactive in UV-Vis range semiconductor material. The TiO₂ nanoparticles differing in size and surface area. Nonetheless, despite different morphology and polymorphic composition, all pristine titanium(IV) oxide particles own wide bandgap energy (Eg), which differs in the range of 3.0–3.2 eV, for rutile and anatase, respectively [7]. In this regard, TiO₂ photoexcitation is possible only with UV irradiation ($\lambda < 388$ nm), and therefore the application of solar radiation is highly limited.



Much effort has been done to shift TiO₂ excitation energy to longer wavelengths, especially in visible light range (400–750 nm). Among various methods [8–10], surface modification with noble and semi-noble metals is the most widely used and effective method. Wysocka et al. [11] obtained mono-(Pt, Ag, and Cu) and bimetal- (Cu/Ag, Ag/Pt, and Cu/Pt) modified TiO₂ photocatalysts, where TiO₂ matrix was commercially available ST01 (fine anatase particles). Metal ions were reduced using the chemical (NaBH₄ solution) as well as thermal treatment methods. Klein et al. [12] prepared TiO₂-P25 modified with Pt, Pd, Ag, and Au using radiolysis reduction. Obtained photocatalysts were further immobilized on the glass plate and used for toluene removal from the gas phase. Moreover, Janczarek et al. [13] proposed a method of obtaining Ag- and Cu-modified decahedral anatase particles (DAP) by photodeposition. Wei et al. [14] reported that selective deposition of nanometals (Au/Ag nanoarticles) on (001) facets of decahedral anatase particles (DAP), together with octahedral anatase particles (OAP) result in significant photocatalytic process improvement under visible light irradiation.

Another possibility of titania visible light activation is the introduction of intrinsic defects to its crystal structure. Titanium or oxygen vacancies and surface disorders led to changes in electronic and crystal changes, resulting in better electrons and holes separation and even bandgap narrowing [15]. Defected TiO₂ is often evidenced by their color: pale blue for oxygen vacancies (due to d–d transitions of bandgap states) and yellow for titanium vacancies (consumption of free electrons and holes), which also suggest shifting its light absorption to the visible light range [15].

Nevertheless, TiO₂ particle size and shape determine not only the number of active sites but also separation and reusability properties. Commercially available titanium(IV) oxide—P25 forms a stable suspension due to its nanometric sizes, which cause the detached process to be highly expensive and energy-consuming [16,17]. The immobilization of nanoparticles on solid substrates [18] could, in turn, result in a decrease in photocatalytic activity due to significantly reducing specific surface area [19,20]. Alternative way of photocatalyst separation is its deposition on magnetic compound, such as Fe₃O₄ [21,22], CoFe₂O₄ [23], ZnFe₂O₄ [24], and BaFe₁₂O₁₉ [25]. Along with using magnetic materials, the percentage of photocatalyst recovery and the possibility of its reuse significantly increase [26–29]. However, the direct contact of ferrite particles with photocatalyst (e.g., TiO₂) may result in unfavorable electron transfer from TiO₂ into the magnetic compound, causing its transformation and photocorrosion [26]. Thus, their separation with an inert interlayer of silica [26–31] or carbon [32] could effectively prevent the charge carriers recombination. In this regard, the magnetic Fe₃O₄@SiO₂/TiO₂ nanocomposites with a core-shell structure, where the core was Fe₃O₄, the photoactive shell was TiO₂, and silica was used as an inert interlayer, are a relatively new and promising group of composite materials.

The previous studies focused on the preparation and characterization of noble metal NPs modified with different titania matrices [33,34], as well as TiO_2 nanoparticles deposited on various magnetic cores (Fe₃O₄, CoFe₂O₃, and ZnFe₂O₄) [26–29]. However, in the literature, there is a lack of complex researches on the correlation between structural defects and TiO_2 modification with different materials (metal nanoparticles or other oxides).

In this regard, deeply characterized defective d-TiO₂ with stable titanium vacancies, obtained by a simple hydrothermal method was further modified with Pt and Cu nanoparticles as well as deposited on the magnetite core. The obtained samples were characterized by X-ray diffractometry (XRD), specific surface area (BET) measurements, UV-Vis diffuse reflectance spectroscopy (DR-UV/Vis), X-ray photoelectron spectroscopy (XPS), and transmission electron microscopy (TEM). The photodegradation of phenol as a model organic pollutant in the presence of the obtained photocatalysts was subsequently investigated in the range of UV-Vis and Vis irradiation. Further, the mechanism of phenol degradation and the role of four oxidative species (h^+ , e^- , $^{\bullet}OH$, and $^{\bullet}O_2^-$) in the studied photocatalytic process were investigated.

2. Results

2.1. Physicochemical Characterization of d-TiO₂-Pt/Cu and Magnetic Fe₃O₄@SiO₂/d-TiO₂-Pt/Cu Photocatalysts

The preparation of d-TiO₂ photocatalysts was based on the hydrothermal method in the oxidative environment (addition of 20–75 mol% of HIO₃). The electron paramagnetic resonance (EPR) analysis confirmed the presence of titanium vacancies with g value of 1.995 after calcination. The signal at EPR spectra for bulk Ti^{3+} was not observed for pure TiO₂ sample, see in Figure 1.

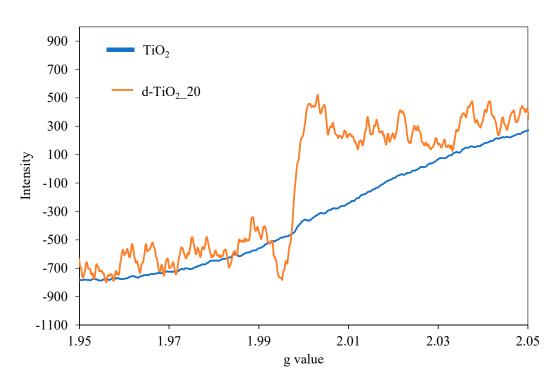


Figure 1. The electron paramagnetic resonance (EPR) spectra in the room temperature for TiO_2 (blue line) and TiO_2 with titania vacancies (orange line).

The general physicochemical and photocatalytic characteristics of the obtained defective $d-TiO_2-Pt/Cu$ and $Fe_3O_4@SiO_2/d-TiO_2-Pt/Cu$ samples, i.e., BET surface area, pore volume, calculated bandgap (Eg) and phenol degradation efficiency in UV-Vis and Vis light are presented in Tables 1 and 2.

Sample	BET (m ² ·g ⁻¹)	V Pores (cm ³ ·g ⁻¹)	Eg (eV)	Rate Constant k (min ⁻¹)·10 ⁻² UV-Vis	Phenol Removal (%) UV-Vis	Rate Constant k (min ^{−1})·10 ^{−2} Vis	Phenol Removal (%) Vis
TiO ₂	169	0.0836	3.2	0.98	46	0.03	3
d-TiO2_20	172	0.0847	2.7	0.79	40	0.17	13
d-TiO ₂ _75	167	0.0826	2.9	0.52	29	0.03	4
TiO ₂ -Pt0.05	166	0.0819	3.2	2.30	76	0.22	14
d-TiO2_20-Pt0.05	148	0.0727	2.85	1.10	48	0.33	20
d-TiO ₂ _75-Pt0.05	164	0.0808	2.9	1.02	46	0.26	15
d-TiO2_20-Pt0.1	152	0.0747	2.7	1.42	57	0.41	22
d-TiO ₂ _20-Cu0.1	101	0.0499	2.9	0.51	30	0.12	7
d-TiO ₂ _20-Pt0.1/Cu0.1	152	0.0744	2.75	1.47	59	0.27	17

Table 1. Physicochemical and photocatalytic characteristics of the obtained defective d-TiO₂-Pt/Cu samples.

Table 2. Physicochemical and photocatalytic characteristics of the obtained magnetic Fe₃O₄@SiO₂/d-TiO₂-Pt/Cu samples.

Sample	BET (m ² ·g ⁻¹)	V Pores (cm ³ ·g ⁻¹)	Rate Constant k (min ^{−1})·10 ^{−2} UV-Vis	Phenol Removal (%) UV-Vis	Rate Constant k (min ^{−1})·10 ^{−2} Vis	Phenol Removal (%) Vis
Fe ₃ O ₄ @SiO ₂ /d-TiO ₂ _20	117	0.0578	0.41	20	0.13	8
Fe ₃ O ₄ @SiO ₂ /d-TiO ₂ _20-Pt0.05	115	0.0568	0.15	11	0.08	4
Fe ₃ O ₄ @SiO ₂ /d-TiO ₂ _20-Pt0.1	122	0.0602	0.41	19	0.19	11
Fe ₃ O ₄ @SiO ₂ /d-TiO ₂ _20-Cu0.1	117	0.0579	0.22	15	0.05	2
$Fe_{3}O_{4}@SiO_{2}/d\text{-}TiO_{2}_20\text{-}Pt0.1/Cu0.1$	117	0.0580	0.36	22	0.26	11

The BET surface area of pure TiO₂ obtained from Titanium(IV) butoxide (TBT) hydrolysis in water and defective TiO₂ samples was similar and ranged from 167 to 172 m²·g⁻¹. The specific surface area of the metal-modified d-TiO₂ samples fluctuated from 166 to 101 m²·g⁻¹ and depended on the type and amount of metallic species deposited on d-TiO₂ surface. The samples modified with Pt NPs revealed a higher BET surface area of about 148 m²·g⁻¹ compared to d-TiO₂ modified with copper oxide (101 m²·g⁻¹), and bimetallic Pt/Cu NPs (152 m²·g⁻¹). The relations between photoactivity in UV-Vis and Vis light range versus BET surface area are also shown in Tables 1 and 2. The obtained results indicated that not so much the surface area but rather the presence of Ti defects and modification with metal nanoparticles caused the enhanced photoactivity of the obtained photocatalysts. Moreover, as shown in Table 2, the addition of surface-modifying metal nanoparticles, as well as further deposition of d-TiO₂-Pt/Cu on magnetic matrice, did not affect the magnitude order of the BET surface area, which remained in the range of 101 to 172 m²·g⁻¹ for d-TiO₂-20-Cu0.1 and d-TiO₂-20, respectively.

The energy bandgaps for all samples were calculated from the plot of $(Kubelka-Munk \cdot E)^{0.5}$ versus E, where E is energy equal to hv, and summarized in Table 1. The samples consist of defective TiO₂ exhibited narrower bandgap of 2.7–2.9 eV compared to TiO₂ and TiO₂-Pt0.05 photocatalysts. Moreover, for all metal-modified defective photocatalysts, the bandgap value, calculated from Kubelka–Munk, transformation did not change, compared to d-TiO₂ matrice, indicating surface modification than doping [11].

The XRD patterns for selected d-TiO₂-Pt/Cu and Fe₃O₄@SiO₂/d-TiO₂-Pt/Cu samples are presented in Figures 1 and 2, with a detailed phase composition and crystalline sizes for all photocatalysts being listed in Tables 3 and 4. Peaks marked "A", "R", and "B" corresponds to anatase, rutile, and brookite phases, respectively. Both crystalline structures (anatase and brookite) appeared for pure TiO₂ prepared by the sol–gel method. For Pt-modified TiO₂ anatase was the major phase, whereas brookite existed as the minor phase. The average crystallite size of anatase was 5–6 nm. The preparation of d-TiO₂ photocatalysts proceeded in the oxidative environment. The introduction into the crystal structure of various types of defects promotes the transformation of anatase to rutile at lower temperatures. Therefore, for the samples obtained in the presence of HIO₃ as the oxidizing agent, after the annealing process the percentage of anatase (the most intense peak at $25^{\circ} 2\theta$, with the (101) plane diffraction, ICDD's card No. 7206075) was decreased in favor of (110) rutile, with the peak at 31° 20 (ICDD's card No. 9004141), even below the anatase to rutile phase transformation temperature [35–37]. For the samples d-TiO₂_75 and d-TiO₂_75-Pt0.05, the dominant phase was rutile with a crystallite size of about 6 nm. Further, the surface modification with plasmonic platinum and semi-noble copper did not cause changes in anatase crystallite size, remaining about 5-6 nm size. The percentage of the brookite phase increased to 8.5% and 13% for d-TiO₂ _20-Pt0.1/Cu0.1, and d-TiO₂_20-Pt0.1 samples, respectively. It resulted from the additional thermal treatment after metal nanoparticles deposition on the photocatalyst surface. Moreover, Pt and Cu modification of TiO₂ did not cause the shift of the peaks in the XRD pattern. The presence of platinum and copper deposited on TiO_2 was not approved (no peaks for platinum or copper) due to low content (0.05–0.1 mol%) and nanometric size.

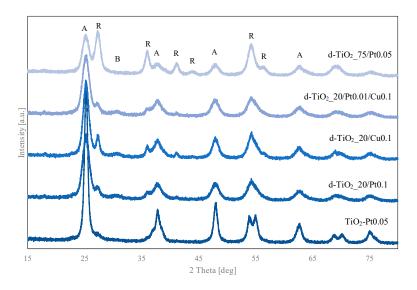


Figure 2. X-ray diffractometry (XRD) patterns for selected defective d-TiO₂-Pt/Cu photocatalysts (A—anatase, B—brookite, and R—rutile).

			Crystalline Si	ze and Phase Content		
Sample	Anatase		Rutile		Brookite	
	Size (nm)	Phase Content (wt.%)	Size (nm)	Phase Content (wt.%)	Size (nm)	Phase Content (wt.%)
TiO ₂	5.97 ± 0.04	95.5 ± 1	-	-	6.1 ± 0.3	4.5 ± 1
d-TiO ₂ _20	5.14 ± 0.03	96 ± 1.0	-	-	4.0 ± 0.6	3.5 ± 0.5
d-TiO ₂ _75	5.67 ± 0.05	21 ± 3.5	6.57 ± 0.09	80 ± 2	-	-
TiO ₂ -Pt0.05	5.71 ± 0.06	91 ± 0.5	-	-	5.7 ± 0.6	9 ± 1
d-TiO ₂ _20-Pt0.05	5.66 ± 0.03	85 ± 8	9.8 ± 0.7	9 ± 1.0	4.9 ± 0.3	6 ± 0.5
d-TiO ₂ _75-Pt0.05	5.49 ± 0.04	48 ± 2	7.53 ± 0.12	52 ± 2	-	-
d-TiO ₂ _20-Pt0.1	5.58 ± 0.03	81 ± 8	7.6 ± 0.8	6 ± 1	4.8 ± 0.3	13 ± 1
d-TiO ₂ _20-Cu0.1	6.62 ± 0.04	72 ± 11	11.7 ± 0.4	9 ± 2	1.52 ± 0.08	8.5 ± 1
d-TiO ₂ _20-Pt0.1/Cu0.1	5.52 ± 0.03	83 ± 12	9.5 ± 0.6	8 ± 2	5.1 ± 0.3	8.5 ± 1

Table 3. Crystalline phases characteristic for obtained defective d-TiO ₂ -Pt/Cu	samples.

Table 4. Crystalline phases characteristic for the obtained magnetic Fe₃O₄@SiO₂/d-TiO₂_20-Pt/Cu samples.

			Crystalline Si	ze and Phase Content		
Sample	Anatase		Rutile		Magnetite	
	Size (nm)	Phase Content (wt.%)	Size (nm)	Phase Content (wt.%)	Size (nm)	Phase Content (wt.%)
Fe ₃ O ₄ @SiO ₂ /d-TiO ₂ _20	5.19 ± 0.05	71 ± 1.5	8.6 ± 0.5	8 ± 0.5	46.1 ± 1.1	21 ± 0.5
Fe ₃ O ₄ @SiO ₂ /d-TiO ₂ _20-Pt0.05	5.60 ± 0.05	71 ± 1.5	8.9 ± 0.5	7 ± 0.5	45.7 ± 1.4	21 ± 0.5
Fe ₃ O ₄ @SiO ₂ /d-TiO ₂ _20-Pt0.1	5.49 ± 0.05	68 ± 2	9.1 ± 0.6	7 ± 1	47.2 ± 4.0	24 ± 0.5
Fe ₃ O ₄ @SiO ₂ /d-TiO ₂ _20-Cu0.1	7.81 ± 0.17	57 ± 2	13.3 ± 1.4	5 ± 1	37.1 ± 1.8	28 ± 1.5
Fe ₃ O ₄ @SiO ₂ /d-TiO ₂ _20-Pt0.1/Cu0.1	5.48 ± 0.05	69 ± 1	7.9 ± 0.4	8 ± 0.5	42.6 ± 3.3	22 ± 1

The XRD analysis of Fe₃O₄@SiO₂/d-TiO₂-Pt/Cu confirmed the formation of a magnetic composite, and, as observed in Figure 3 and Table 4, there was no significant difference between the diffraction patterns of the obtained magnetic photocatalysts modified with Pt/Cu NPs. The presence of pure magnetite, with diffraction peaks at 30.2° , 35.6° , 43.3° , 57.3° , and 62.9° 20 corresponding to (220), (311), (400), (511), and (440) cubic inverse spinel planes (ICDD's card No. 9005813) was confirmed for all Fe₃O₄@SiO₂/d-TiO₂-Pt/Cu magnetic photocatalysts. The decrease in Fe₃O₄ peaks intensity was caused by the formation of tight non-magnetic shell on the core surface, which was previously described by Zielińska-Jurek et al. [26]. The broad peak at 15–25° 20 corresponds to amorphous silica [38,39]. The content of the magnetite crystalline phase varied from 21% to 28% for Fe₃O₄@SiO₂/d-TiO₂_20-Pt0.05 and Fe₃O₄@SiO₂/d-TiO₂_20-Cu0.1, respectively. At the same time, TiO₂ crystallite size and anatase to rutile phase content ratio remained unchanged for Fe₃O₄@SiO₂/d-TiO₂_20 and Fe₃O₄@SiO₂/d-TiO₂_20-Pt/Cu samples. No other crystalline phases were identified in the XRD patterns, which indicated the crystal purity of the obtained composites.

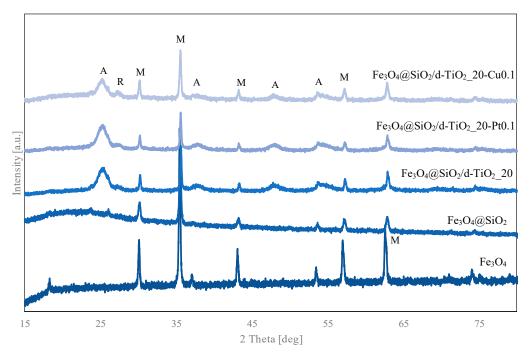


Figure 3. XRD patterns of magnetic photocatalysts, compared with Fe₃O₄ and Fe₃O₄@SiO₂ (A—anatase, R—rutile, and M—magnetite).

The photoabsorption properties of metal-modified defective d-TiO₂ samples were studied by diffuse reflectance spectroscopy, and exemplary data are shown in Figure 4. Comparing to pure TiO₂ photocatalyst, introducing platinum as a surface modifier caused an increase of absorption in the visible light region, however, without shifting a maximum, as presented for sample TiO₂–Pt0.05. Modification of defective d-TiO₂ with Pt and Cu was associated with a further increase of Vis light absorbance and proportional to the amount of the deposited metal. Moreover, the deposition of Pt caused a more significant absorbance increment than the same modification with Cu species.

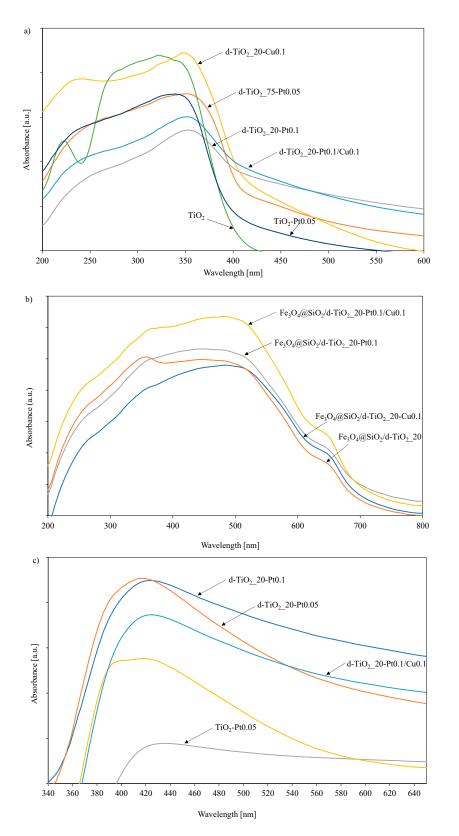


Figure 4. DR/UV-Vis spectra for selected d-TiO₂-Pt/Cu (**a**) and Fe₃O₄@SiO₂/d-TiO2-Pt/Cu (**b**) photocatalysts together with Pt and Cu plasmon determination with bare TiO₂ as a reference (**c**).

Defective d-TiO₂-Pt/Cu deposited on $Fe_3O_4@SiO_2$ core were characterized by extended light absorption ranged to 700 nm. It could be observed that the described absorption properties in the Vis light for metal-modified TiO₂ and absorption properties of final composites have been preserved.

The presence of Localized Surface Plasmon Resonance (LSPR) peaks for Pt and Cu were confirmed based on DR-UV/Vis spectra measurements with pure TiO_2 as a reference (see in Figure 4c). Platinum surface plasmon resonance was observed at the wavelength of about 410–420 nm [33,40]. Electron transfer between Cu(II) and valence band of titanium(IV) oxide could be confirmed by absorption increment from 400 to 450 nm. The typical LSPR signal for zero valent copper at 500–580 nm was not observed, suggesting that Cu is mainly present in its oxidized forms [41,42].

To confirm the presence of noble metal and semi-noble metal NPs on defective TiO₂ surface, the XPS analyses for the selected photocatalysts and deconvolution of Pt 4f and Cu 2p were performed, and the results are presented in Figure 5. Platinum species deposited on the titania surface were designated by deconvolution of Pt 4f peak into two components: Pt 4f_{7/2} and Pt 4f_{5/2}. According to the literature, Pt 4f_{7/2} peak, with binding energies in the range of 74.2 to 75.0 eV, refers to the Pt⁰, while Pt 4f_{5/2} peak, appearing at 77.5–77.9 eV is assigned to Pt⁴⁺ [11]. The main peaks for Cu 2p appeared as Cu $2p_{3/2}$ and Cu $2p_{1/2}$ at 934 eV and 952 eV. Both of those peaks are commonly attributed to Cu⁺ and Cu²⁺ ions [13,43,44]. Obtained data indicated that both Pt and Cu species were successfully deposited on the titania surface.

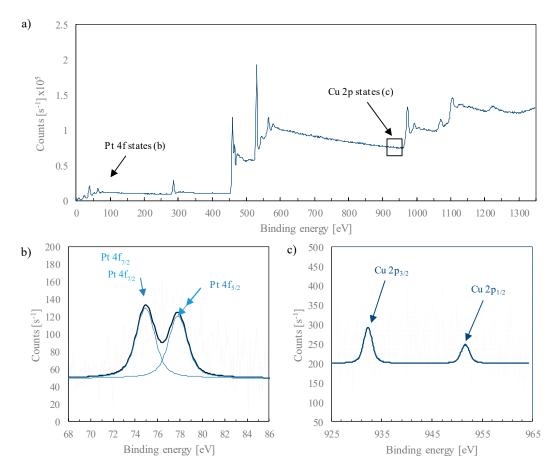


Figure 5. XPS spectra for d-TiO₂_20-Pt0.1/Cu0.1 sample (**a**) with the deconvolution for Pt 4f (**b**) and Cu 2p (**c**).

Moreover, the presence of Pt NPs at the surface of the magnetic nanocomposites was also confirmed by microscopy analysis. As presented in Figure 6, the formation of SiO_2/TiO_2 shell, with a thickness of about 20 nm, tightly covering magnetite nanoparticles was observed. Platinum nanoparticles with a diameter of about 10–20 nm were evenly distributed on the d-TiO₂ layer.

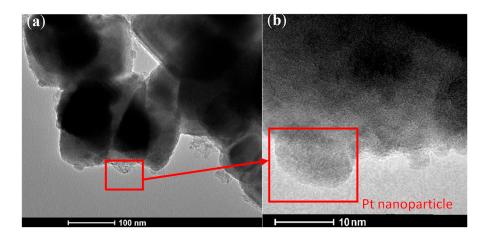


Figure 6. Transmission electron microscopy (TEM) analysis for $Fe_3O_4@SiO_2/d$ -Ti $O_2_20/Pt0.1$ sample (a) magnification on Pt nanoparticle (b).

2.2. Photocatalytic Activity of d-TiO₂-Pt/Cu and Fe₃O₄@SiO₂/d-TiO₂-Pt/Cu Photocatalysts

The effect of Pt and Cu presence on the properties of defective d-TiO₂ photocatalysts was evaluated in reaction of phenol degradation under UV-Vis and Vis light irradiation. The results, presented as the efficiency of phenol degradation as well as phenol degradation rate constant k, are given in Figures 7 and 8. Additionally, the effect of the electron (e⁻), hole (h⁺), hydroxyl radical ($^{\circ}$ OH), and superoxide radical ($^{\circ}$ O₂⁻) scavengers were investigated and presented in Figure 9.

Among analyzed metal-modified photocatalysts, TiO₂–Pt0.05 revealed the highest phenol degradation in UV-Vis light. After 60 min of irradiation, about 76% of phenol was degraded. After introducing plasmonic platinum and semi-noble copper species as a surface modifiers, UV-Vis photoactivity of defective d-TiO₂ samples increased to 59%. The degradation rate constant k increased to 1.47×10^{-2} min⁻¹ compared to d-TiO₂_20 (0.79×10^{-2} min⁻¹), and d-TiO₂_75 (0.52×10^{-2} min⁻¹) photocatalysts. Nonetheless, the most significant changes were observed during the photocatalytic process in visible light ($\lambda > 420$ nm). Modifying with 0.05 mol% of Pt, the surface of almost inactive in Vis light d-TiO₂_75 resulted in three-times higher photocatalytic activity under visible light. Therefore, a highly positive effect of metal surface modification of defective d-TiO₂ photocatalysts surface was noticed. It resulted from better charge carriers' separation and decreasing the electron-hole recombination rate. Moreover, the narrower bandgap of the defective d-TiO₂ (in comparison with pure TiO₂) and modification with Pt possessing surface plasmon resonance properties, could also enhance visible light absorption and consequently led to photocatalytic activity increase.

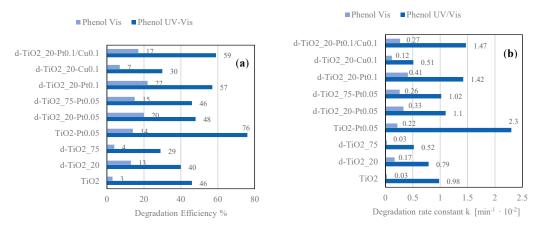


Figure 7. Efficiency of phenol degradation in UV-Vis and Vis light for d-TiO₂-Pt/Cu photocatalysts, presented as % of degradation (**a**) and rate constant k (**b**).

Pure magnetite, coated with inert silica, did not affect the photocatalytic process. Furthermore, the $Fe_3O_4@SiO_2/d$ -TiO_2 composite modified with Pt NPs, and bimetallic Pt/Cu NPs revealed the highest photocatalytic activity in Vis light range. The phenol degradation rate constant in Vis light was 2-times higher for $Fe_3O_4@SiO_2/d$ -TiO_2-Pt/Cu compared to $Fe_3O_4@SiO_2/d$ -TiO_2 sample. However, the obtained magnetic photocatalysts had similar photocatalytic activity in UV-Vis light, almost regardless of the surface modification of d-TiO_2 with noble metals. It probably resulted from larger Pt particles (~20 nm) deposition at the surface of $Fe_3O_4@SiO_2/d$ -TiO_2 composite than for TiO_2–Pt0.05 with particles size of about 2–3 nm. Previously, we have reported that the size of noble metal nanoparticles, especially platinum, deposited on the TiO_2 surface strictly depends on the semiconductor surface area, as well as its crystal lattice defects [33,45]. Fine metal particles are produced on the TiO_2 surface with a developed specific surface area with a high density of oxygen traps and nucleation sites, and the highest photocatalytic activity is noticed for Pt-modified photocatalyst, where the size of Pt is below 3 nm [33]. In the present study, Pt nanoparticles' average diameter was about 20 nm as a result of the deposition of Pt ions and their reduction on formed particles' defects. Therefore, the lower metal/semiconductor interface resulted in a decrease in photocatalytic activity under UV-Vis light irradiation.

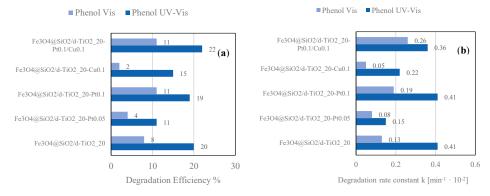


Figure 8. Efficiency of phenol degradation in UV-Vis and Vis light for magnetic $Fe_3O_4@SiO_2/d$ -TiO_2-Pt/Cu photocatalysts, presented as % of degradation (**a**) and rate constant k (**b**).

For the final stability and reusability test, the most active defective photocatalyst was selected. For sample d-TiO₂_20/Pt0.1/Cu0.1, three 1-h-long subsequent cycles of phenol degradation under UV-Vis light were performed. The obtained results are presented in Figure 9.

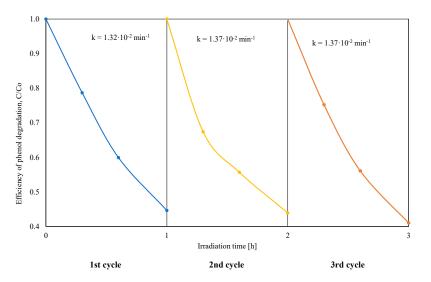


Figure 9. Efficiency of UV-Vis phenol degradation in the presence of defective d-TiO₂_20/Pt0.1/Cu0.1 photocatalyst measured in the three subsequent cycles.

There was no significant change in phenol degradation rate constant after the second and third cycles. Thus, the analyzed photocatalyst revealed good stability and reusability.

Furthermore, the reactive species were investigated to understand the photocatalytic reaction mechanism. Benzoquinone (BQ), silver nitrate (SN), ammonium oxalate (AO), and tert-butanol (t-BuOH) were used as superoxide radical anions (\cdot O₂⁻), electrons (e⁻), holes (h⁺), and hydroxyl radicals (\cdot OH) scavengers, respectively. Obtained results, presented as phenol degradation rate constant k, in comparison to the photodegradation process without scavengers, are presented in Figure 10. The most significant impact on phenol degradation reaction in the presence of metal-modified d-TiO₂ was observed for superoxide radicals. After introducing to the photocatalyst suspension BQ solution, the phenol degradation efficiency was significantly inhibited. A slight decrease was also observed in the presence of SN as an electron trap. On the other hand, the addition of AO and t-BuOH did not decrease the phenol degradation rate.

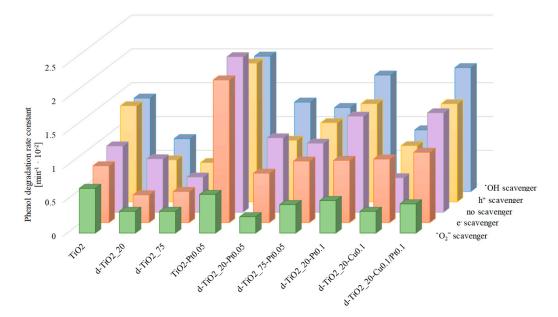


Figure 10. Photocatalytic degradation of phenol for defective $Fe_3O_4@SiO_2/d$ -Ti O_2 -Pt/Cu photocatalysts in the presence of e^- , h^+ , $\bullet O_2^-$, and $\bullet OH$ scavengers.

Modification of TiO₂ resulted in the shift of the valence band as was revealed from the analysis of Mott–Schottky plot, where the relation between applied potential vs. Csc^{-2} is presented (see in Figure 11). According to the intersection with E axis the flat band potential was estimated. In the case of pure titania it equals to -1.2 V, whereas for d-TiO₂_20-Pt0 the value of -1.13 was reached. In order to prepare energy diagram of both materials given in Figure 12, the values of bandgap energy was taken into account. As could be seen, for the modified material the position both the conduction and valence band are shifted. According to Monga et al. [46] the Schottky barrier formed at the metal-TiO₂ interface affecting the efficiency of e- transfer. The lowering of the CB band edge is in accordance with the literature indicating that the work function of the metal prone decrease of the CB location. Then, the Schotky barrier is decreased at the metal/semiconductor heterojunction. As a result, the transfer of the photoexcited electron from metal NPs to titania is facilitated and plays important role in photocatalytic activity improvement. The introduction of titanium defects to the TiO₂ crystal structure also resulted in narrowing the bandgap from 3.2 to 2.7 eV.

Based on the presented results, a schematic mechanism of UV-Vis phenol degradation in the presence of metal-modified defective $Fe_3O_4@SiO_2/d-TiO_2-Pt/Cu$ photocatalyst was proposed and shown in Figure 12. After irradiation of the photocatalyst surface with UV-Vis light, electrons from the Pt are injected to the conduction band of titania and then utilized in oxygen reduction to form

reactive oxygen radicals. The path of phenol degradation led through several intermediates, such as benzoquinone, hydroquinone, catechol, resorcinol, oxalic acid, and finally, to complete mineralization to CO_2 and H_2O [47–49]. An analysis of possible charge carriers' impact revealed that for photoactivity of d-TiO₂-Pt/Cu, they are responsible for mainly generated superoxide radicals. The phenol degradation mechanism proceeded by the generation of reactive oxygen species, e.g., ${}^{\bullet}O_2{}^{-}$, which attacked the phenol ring, resulting in benzoquinone and hydroquinone formation confirmed by high-performance liquid chromatography (HPLC) analyses. Moreover, during the photoreaction, the concentration of formed intermediates decreased, which suggests mineralization of recalcitrant chemicals to simple organic compounds.

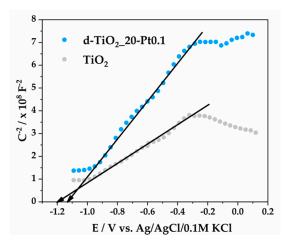


Figure 11. The Mott–Schottky plot for the bare and Pt modified d-TiO₂.

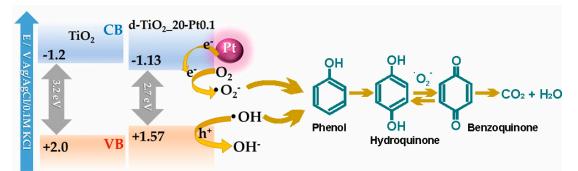


Figure 12. Energy diagram depicting the position of valence and conduction bands of bare and modified titania including the indication of charge transfer within the Schottky junction, and schematic illustration of phenol degradation mechanism over defective Fe₃O₄@SiO₂/d-TiO₂-Pt/Cu photocatalysts.

3. Materials and Methods

Titanium(IV) oxide organic precursor: titanium(IV) butoxide (>99%) was provided by Alfa Aesar (Haverhill, MA, USA). Iodic acid (99.5%), sodium borohydride (99%), chloroplatinate acid hydrate ($H_2PtCl_4 \cdot xH_2O$) (99.9%) and copper nitrate trihydrate ($Cu(NO_3)_2 \cdot 3H_2O$) (99–104%), used for TiO₂ structure and surface modification, were purchased from Sigma (Poznan, Poland). Ferrous ferric oxide (Fe₃O₄, 97%) with a declared particles size of 50 nm was purchased from Aldrich (Poznan, Poland). Tetraethyl orthosilicate (TEOS) was provided by Aldrich and was used as a precursor for the inert interlayer of magnetic nanoparticles. Ammonium hydroxide solution (25%) was purchased from Avantor (Gliwice, Poland). Chemicals for w/o microemulsions preparation, such as cyclohexane and 2-propanol, were purchased from Avantor. Cationic surfactant, hexadecyltrimethylammonium bromide (CTAB), was provided by Sigma Aldrich (Poznan, Poland). Acetonitrile and orthophosphoric acid (85%) for HPLC mobile phase preparation were provided by Merck (Darmstadt, Germany)

and VWR (Gdansk, Poland), respectively. Phenol, used as a model organic recalcitrant pollutant in photocatalytic activity measurements, was purchased from VWR. For the titania paste formation polyethylene glycol (PEG) from Sigma Aldrich (Poznan, Poland) was used, while Na₂SO₄ used for electrolyte preparation was purchase from VWR. All reagents were used without further purification.

3.1. Preparation of Defective TiO₂-Pt/Cu Photocatalysts

Defective TiO₂ (marked as d-TiO₂) was obtained by the hydrothermal method assisted with the annealing process. Titanium(IV) butoxide (TBT) and iodic acid (HIO₃) were used as a TiO₂ precursor and oxidizing environment for titanium vacancies formation, respectively. Briefly, the appropriate amount of HIO₃ (presented in Table 5) was dissolved in 80 cm³ of distilled water. After that, 10 cm³ of TBT was added dropwise, and the obtained suspension was stirred for 1 h with magnetic stirring at room temperature. In the next step, the suspension was transferred into a Teflon-lined autoclave for thermal treatment at 110 °C for 24 h. The resultant precipitate was centrifuged, dried at 70 °C and then calcined at 300 °C for 3 h.

Table 5. HIO₃ concentrations used for the synthesis of each d-TiO₂ photocatalyst.

Sample	Oxidant Concentration [mol%]	Mass of Added Oxidant [g]
TBT	0	0
d-TiO2_20	20	1.032
d-TiO ₂ _75	75	3.869

The obtained defective d-TiO₂ photocatalysts were modified using platinum and copper nanoparticles by the co-precipitation method. In this regard, d-TiO₂ was dispersed in 50 cm³ of deionized water, and Pt/Cu precursor solutions (0.05 and 0.1 mol% of Pt and 0.1 mol% of Cu with respect to TiO₂) were added. After that, NaBH₄ solution was introduced to reduce the metals ions followed by their deposition on the titania surface. The mole ratio of metal ions to NaBH₄ was 1:3. After the reduction process, the photocatalyst suspension was mixed for 2 h, and the d-TiO₂-Pt/Cu nanoparticles were separated, washed with deionized water, and dried at 80 °C to dry mass. The final step was calcination at 300 °C for 3 h.

3.2. Preparation of Magnetic Fe₃O₄@SiO₂/d-TiO₂-Pt/Cu Nanocomposites

Previously obtained d-TiO₂-Pt/Cu nanoparticles were deposited on a magnetic substrate as a thin photocatalytic active shell. Magnetite (Fe₃O₄) was selected as a core of the designed composite due to its excellent magnetic properties (high Ms value and low Hc), which enable us to separate obtained photocatalyst in the external magnetic field. Silica was used as an interlayer to isolate Fe₃O₄ from TiO₂ and suppress possible electron transfer between them. The magnetic photocatalysts were obtained in the w/o microemulsion system based on changes in the particles surface charge as a function of pH, described in the previous study [26].

Firstly, commercially available Fe_3O_4 nanoparticles with nominate particles diameter of 50 nm were dispersed in water at pH 10. The prepared suspension was then introduced to cyclohexane/isopropanol (100:6 volume ratio) mixture in the presence of cationic surfactant and cetyltrimethylammonium bromide (CTAB) creating stable w/o microemulsion system with water nanodroplets dispersed in the continuous oil phase. The molar ratio of water to surfactant was set at 30. After the microemulsion stabilization, the corresponding amount of tetraethyl orthosilicate (TEOS) was added, resulting in the formation of SiO₂ interlayer on Fe_3O_4 core, after ammonia solution introduced into the microemulsion system. The molar ratio of TEOS to Fe_3O_4 was 8:1, and NH₄OH to TEOS was 16:1. The microemulsion was destabilized using acetone and obtained nanocomposite $Fe_3O_4@SiO_2$ was separated, washed with ethanol and water, dried at 70 °C to dry mass, and calcined at 400 °C for 2 h.

In the second step, previously obtained $Fe_3O_4@SiO_2$ particles were combined with d-TiO₂-Pt/Cu in order to create photocatalytic active nanomaterial. The reversed-phased microemulsion system at

pH 10 was used, and Fe₃O₄ to the TiO₂ molar ratio was equaled to 1:4 [27]. The junction between magnetic/SiO₂ and photocatalytic layers was promoted by their opposite surface charges, provided by the presence of CTAB at the basic conditions. The as-obtained Fe₃O₄@SiO₂/d-TiO₂-Pt/Cu samples, after their separation and purification using water and ethanol, were dried at 70 °C to dry mass and calcined at 300 °C for 2 h.

3.3. Characterization of the Obtained Magnetic Photocatalysts

The XRD analyses were performed using the Rigaku Intelligent X-ray diffraction system SmartLab equipped with a sealed tube X-ray generator (a copper target; operated at 40 kV and 30 mA). Data was collected in the 2 θ range of 5–80° with a scan speed and scan step of 1°·min⁻¹ and 0.01°, respectively. The analyses were based on the International Centre for Diffraction Data (ICDD) databased. The crystallite size of the photocatalysts in the vertical direction to the corresponding lattice plane was determined using Scherrer's equation with Scherrer's constant equal to 0.891. Quantitative analysis, including phase composition with standard deviation, was calculated using the Reference Intensity Ratio (RIR) method from the most intensive independent peak of each phase.

Nitrogen adsorption-desorption isotherms (BET method for the specific surface area) were recorded using the Micromeritics Gemini V (model 2365) (Norcross, GA, USA) instrument at 77 K (liquid nitrogen temperature).

Light absorption properties were measured using diffuse reflectance (DR) spectroscopy in the range of 200–800 nm. The bandgap energy of obtained samples was calculated from $(F(R)\cdot E)^{0.5}$ against E graph, where E is photon energy, and F(R) is Kubelka–Munk function, proportional to the radiation's absorption. The measurements were carried out using ThermoScientific Evolution 220 Spectrophotometer (Waltham, MA, USA) equipped with a PIN-757 integrating sphere. As a reference, BaSO₄ was used.

X-ray photoelectron spectroscopy (XPS) measurements were conducted using Escalab 250Xi multi-spectrometer (Thermofisher Scientific) using Mg K X-rays.

The morphology and distribution size for $Fe_3O_4@SiO_2/d-TiO_2$ -Pt as a reference magnetic nanocomposite sample was further analyzed using HR-TEM imagining (Tecnai F20 X-Twin, FEI Europe) together with elements identification in nanometric scale by EDS mapping.

Electron paramagnetic resonance (EPR) spectroscopy was used for intrinsic defects formation confirmation. Measurements were conducted using RADIOPAN SE/X-2547 spectrometer (Poznań, Poland), operating at room temperature, with frequency in range 8.910984–8.917817 GHz.

3.4. Photocatalytic Activity Analysis

Photocatalytic activity of the obtained samples was evaluated in phenol degradation reaction, both in UV-Vis and Vis light irradiation, using 300 W Xenon lamp (LOT Oriel, Darmstadt, Germany). For the visible light measurements, a cut-off 420 nm filter (Optel, Opole, Poland) was used to obtain a settled irradiation interval. A 0.05 g (2 g·dm⁻³) of a photocatalyst, together with a 20 mg·dm⁻³ phenol solution, was added to a 25 cm³ quartz photoreactor with an exposure layer thickness of 3 cm and obtained suspension was stirred in darkness for 30 min to provide adsorption-desorption equilibrium. After that, photocatalyst suspension was irradiated under continuous stirring and a power flux (irradiation intensity) of 30 mW·cm⁻² for 60 min. The constant temperature of the aqueous phase was kept at 20 °C using a water bath. Every 20 min of irradiation, 1.0 cm³ of suspension was collected and filtered through syringe filters (pore size = $0.2 \mu m$) for the removal of photocatalysts particles. Phenol concentration, as well as a formation of degradation intermediates, were analyzed using reversed-phase high-performance liquid chromatography (HPLC) system, equipped with C18 chromatography column with bound residual silane groups (Phenomenex, model 00F-4435-E0) and a UV-Vis detector with a DAD photodiodes array (model SPD-M20A, Shimadzu). The tests were carried out at 45 °C and under isocratic flow conditions of 0.3 mL·min⁻¹ and volume composition of the mobile phase of 70% acetonitrile, 29.5% water and 0.5% orthophosphoric acid. Qualitative and

quantitative analysis was performed based on previously made measurements of relevant substance standards and using the method of an external calibration curve.

Phenol removal percentage was calculated from the equation:

$$D\% = \frac{C_o - C_n}{C_o} \cdot 100 \tag{1}$$

where: C_0 —phenol initial concentration [mg·dm⁻³], C_n —phenol concentration during photodegradation [mg·dm⁻³].

Rate constant k was determined from $\ln(C_o/C_n)$ against t plot where C_o and C_n are phenol concentrations [mg·dm⁻³] and t is degradation time [min]. Rate constant k is equal to directional coefficient "a" of the plot.

In order to evaluate the stability of the obtained photocatalysts, three 1-h-long subsequent cycles of phenol degradation under UV-Vis light using the most active defective $d-TiO_2_20/Pt0.1/Cu0.1$ sample were performed. After each cycle, photocatalyst was separated from the suspension and use in the next cycle without additional treatment.

The effect of charge carrier scavengers was examined by addition into phenol solution 1 cm^3 of 500 mg·dm⁻³ of tert-butyl alcohol (t-BuOH), benzoquinone (BQ), ammonium oxalate (AO), and silver nitrate (SN), respectively.

3.5. Electrochemical Measurements

In order to prepare Mott–Schottky plot the fabricated titania powders were used to form the paste, deposited using doctor-blade technique onto the Pt support. The paste consist of 0.2 g of photocatalyst in 0.1 g of polyethylene glycol (PEG) and 1 cm³ of deionized water. Finally the calcination was carried out at 400 °C for 5 h with a heating rate 1 °C·min-¹ ensuring removal of the organic binder. The fabricated electrode material stayed as working electrode tested in three electrode arrangement where Ag/AgCl/0.1M KCl and Pt mesh were used as reference and counter electrode, respectively. The deaerated 0.5 M Na₂SO₄ was applied as electrolyte. The electrochemical spectroscopy (EIS) impedance data was recorded from the anodic towards cathodic direction. Prior the tests, the investigated samples were not subjected to any preliminary treatment or measurement and their potential was held to reach a steady-state conditions. EIS data were recorded for the single frequency of 1000 Hz in the potential range from +0.1 to -1.2 V vs. Ag/AgCl/0.1 M KCl using a 10 mV amplitude of the AC signal. The capacitance of space charge layer was further calculated from the imaginary part of the measured impedance following the equation [50]:

$$C_{SC} = \frac{-1}{2\pi f Z_{im}} \tag{2}$$

where f stands for the frequency of the AC signal and Z_{im} for the imaginary part of impedance.

4. Conclusions

Surface modification of defective d-TiO₂ photocatalyst with platinum and copper nanoparticles resulted in a significant increase in its photocatalytic activity, both in UV-Vis and Vis range. The EPR analysis confirmed the presence of Ti defects in the structure of TiO₂ samples. The highest activity in Vis light was noticed for d-TiO₂ modified with Pt NPs. It resulted from surface plasmon resonance properties of Pt and narrowing the bandgap of the defective d-TiO₂. Among magnetic photocatalysts, the highest activity in Vis light was observed for Pt-modified and Pt/Cu-modified defective d-TiO₂ deposited on $Fe_3O_4@SiO_2$ magnetic core. Analysis of phenol degradation mechanism revealed that superoxide radicals are mainly responsible for phenol oxidation and mineralization. However, the photocatalytic activity in reaction of phenol degradation in UV-Vis light in the presence of Pt-modified $Fe_3O_4@SiO_2/d$ -TiO₂ with the Pt particle size of about 20 nm was comparable with

the activity of $Fe_3O_4@SiO_2/d$ -TiO₂. It resulted from the deposition of Pt NPs in the place of titanium vacancies, and as a consequence formation of larger metal particles due to the seed-mediated growth mechanism on the TiO₂. In this regard, a lower metal/semiconductor interface resulted in a decrease in photocatalytic activity in the UV-Vis spectrum range. Furthermore, the creation of a core-shell magnetic $Fe_3O_4@SiO_2/d$ -TiO₂-Pt/Cu nanostructures allowed an effective separation of the obtained magnetic photocatalysts.

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