

Communication

Ag Nanoparticle–Functionalized Open-Ended Freestanding TiO₂ Nanotube Arrays with a Scattering Layer for Improved Energy Conversion Efficiency in Dye-Sensitized Solar Cells

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Abstract: Dye-sensitized solar cells (DSSCs) were fabricated using open-ended freestanding TiO₂ nanotube arrays functionalized with Ag nanoparticles (NPs) in the channel to create a plasmonic effect, and then coated with large TiO₂ NPs to create a scattering effect in order to improve energy conversion efficiency. Compared to closed-ended freestanding TiO₂ nanotube array-based DSSCs without Ag or large TiO₂ NPs, the energy conversion efficiency of closed-ended DSSCs improved by 9.21% (actual efficiency, from 5.86% to 6.40%) with Ag NPs, 6.48% (actual efficiency, from 5.86% to 6.24%) with TiO₂ NPs, and 14.50% (actual efficiency, from 5.86% to 6.71%) with both Ag NPs and TiO₂ NPs. By introducing Ag NPs and/or large TiO₂ NPs to open-ended freestanding TiO₂ nanotube array-based DSSCs, the energy conversion efficiency was improved by 9.15% (actual efficiency, from 6.12% to 6.68%) with Ag NPs and 8.17% (actual efficiency, from 6.12% to 6.62%) with TiO₂ NPs, and by 15.20% (actual efficiency, from 6.12% to 7.05%) with both Ag NPs and TiO₂ NPs. Moreover, compared to closed-ended freestanding TiO₂ nanotube arrays, the energy conversion efficiency of open-ended freestanding TiO₂ nanotube arrays increased from 6.71% to 7.05%. We demonstrate that each component—Ag NPs, TiO₂ NPs, and open-ended freestanding TiO₂ nanotube arrays—enhanced the energy conversion efficiency, and the use of a combination of all components in DSSCs resulted in the highest energy conversion efficiency.

Keywords: open-ended freestanding TiO₂ nanotube arrays; dye-sensitized solar cells; plasmonic; scattering; anodization

1. Introduction

Since the original work by O'Regan and Grätzel in 1991 [1], dye-sensitized solar cells (DSSCs) have been investigated extensively because of their high energy conversion efficiency and low cost [2–9]. Generally, mesoporous TiO₂ nanoparticle (NP) films and ruthenium sensitizers are used for DSSCs [2–4,10–16]. However, the efficiency of mesoporous TiO₂ NP film-based DSSCs is limited by grain boundaries, defects, and numerous trapping sites. Moreover, mesoporous TiO₂ NP films can cause charge recombination and mobility [17,18].

TiO₂ nanotubes, which enhance electron transport and charge separation by creating direct pathways and accelerating charge transfer between interfaces, have great potential to overcome the

problems of mesoporous TiO₂ NP films [19–22]. TiO₂ nanotubes can be prepared by a hydrothermal method [23] or an electrochemical method [24], known as anodization. TiO₂ nanotube arrays prepared by anodization have a well-ordered and vertically oriented tubular structure that facilitates a high degree of electron transport and less charge recombination than mesoporous TiO₂ NP films [25–27]. There is much room for improvement in the energy conversion efficiency of current DSSCs based on TiO₂ nanotube arrays compared to the relatively extensively researched mesoporous TiO₂ NP film-based DSSCs [28].

To date, several approaches for improving the energy conversion efficiency of TiO₂ nanotube array-based DSSCs have been reported. Metal NPs, which can harvest light via surface plasmon resonance (SPR), have been used to enhance the energy conversion efficiency of DSSCs by introducing Au or Ag NPs into TiO₂ nanotube arrays [29–32]. Barrier layers remove TiO₂ nanotube arrays, so open-ended TiO₂ nanotube arrays, which can also be classified as arrays of columnar nanopores, have been used for DSSCs to provide increased energy conversion efficiency [33]. Moreover, the energy conversion efficiency of TiO₂ nanotube array-based DSSCs can be further increased by introducing a scattering layer to the active layer [34].

So far, TiO₂ nanotubes that make use of a scattering layer [34] or plasmonic materials [14] have been reported, but a scattering layer with plasmonic materials has not been used in TiO₂ nanotube-based DSSCs. In this study, we report the development of freestanding TiO₂ nanotube arrays filled with Ag NPs and large TiO₂ NPs, which improve the energy conversion efficiency of DSSCs. Furthermore, we compare the effects of Ag NPs and large TiO₂ NPs in open- and closed-ended freestanding TiO₂ nanotube arrays in DSSCs. The energy conversion efficiencies of the following eight types of DSSCs were compared: closed-ended freestanding TiO₂ nanotube arrays with/without Ag NPs and/or a TiO₂ scattering layer and open-ended freestanding TiO₂ nanotube arrays with/without Ag NPs and/or a TiO₂ scattering layer.

2. Results and Discussion

2.1. Structure of DSSCs with Freestanding TiO₂ Nanotube Arrays with Channels Containing Ag NPs

Figure 1 illustrates the fabrication of DSSCs with Ag NPs and large TiO₂ NPs to enable plasmonic and scattering effects in open-ended freestanding TiO₂ nanotube array-based DSSCs. Ti plates were anodized and then annealed at 500 °C for 1 h to prepare anatase TiO₂ nanotube arrays. After carrying out secondary anodization, the TiO₂ nanotube arrays were easily detached from the Ti plates. TiO₂ nanotube arrays, once separated from the Ti plates, are termed “closed-ended freestanding TiO₂ nanotube arrays”. Freestanding TiO₂ nanotube arrays have a barrier layer at the bottom that disturbs electron transport and electrolyte diffusion. This barrier layer was removed using the ion-milling method with several minutes of Ar⁺ bombardment to yield “open-ended freestanding TiO₂ nanotube arrays”. The closed- and open-ended freestanding TiO₂ nanotube arrays were transferred to fluorine-doped tin oxide (FTO) glass using TiO₂ paste and annealed to enhance the adhesion between the closed- and open-ended freestanding TiO₂ nanotube arrays and the fluorine-doped tin oxide (FTO) glass. To improve the energy conversion efficiency by the plasmonic effect, Ag NPs were embedded in the channel of freestanding TiO₂ nanotube arrays using 254 nm ultraviolet (UV) irradiation with aqueous silver nitrate. To further enhance the energy conversion efficiency, large TiO₂ NPs (400 nm) as a scattering layer were coated onto the active layer by the doctor blade method. This substrate was sandwiched with the counter electrode and filled with electrolyte. The active area of the DSSCs was ~0.25 cm².

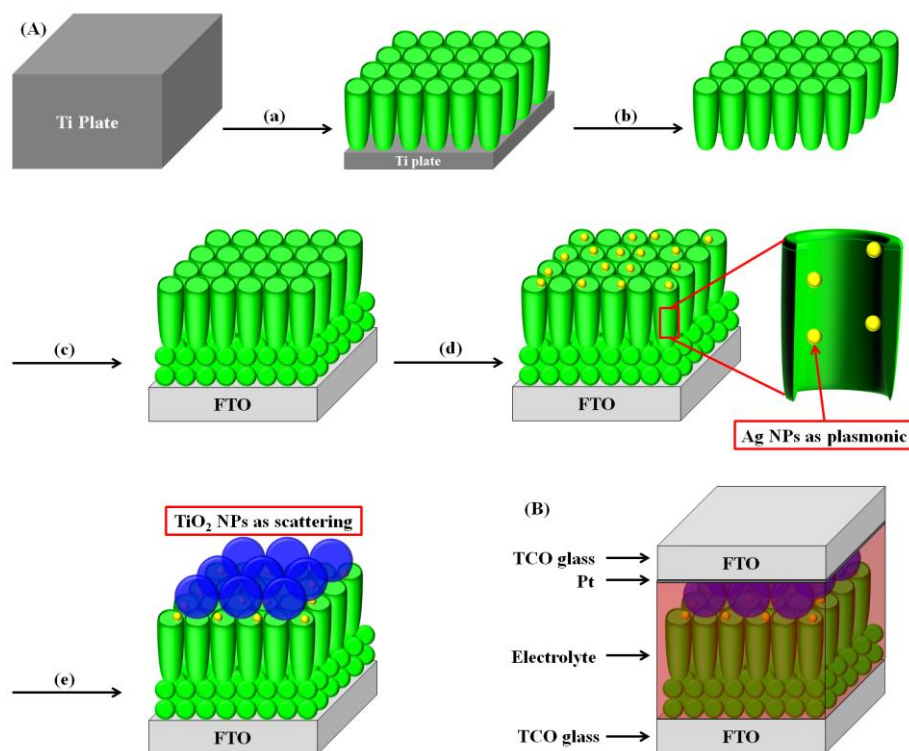


Figure 1. Overall scheme of dye-sensitized solar cells (DSSCs) with open-ended freestanding TiO₂ nanotube arrays with Ag nanoparticles (NPs) and large TiO₂ NPs. (A) (a) Ti anodization for TiO₂ nanotube arrays; (b) freestanding TiO₂ nanotube arrays and etching by ion milling; (c) transference of open-ended freestanding TiO₂ nanotube arrays onto fluorine-doped tin oxide (FTO) glass; (d) formation of Ag NPs by ultraviolet (UV) irradiation; and (e) introduction of large TiO₂ NPs. (B) Structure of a DSSC with freestanding TiO₂ nanotube arrays and large TiO₂ NPs.

2.2. Characterization of Freestanding TiO₂ Nanotube Arrays with Channels Containing Ag NPs

Field emission scanning electron microscope (FE-SEM) images of freestanding TiO₂ nanotube (TNT) arrays are shown in Figure 2. The top side of the freestanding TiO₂ nanotube arrays had 100-nm-diameter pores, as shown in Figure 2a. The bottom layer of closed-ended freestanding TiO₂ nanotube arrays before ion milling lacked pores, as shown in Figure 2b. However, after ion milling, 20-nm-diameter pores were evident on the bottom layer of open-ended freestanding TiO₂ nanotube arrays, as shown in Figure 2c. Open-ended TNT arrays can be prepared by chemical etching [35] or physical etching [33,36]. In the chemical etching method, the bottom layers of TNT arrays were easily removed by the etchant. However, the surface morphology and length of TNT arrays were also dissolved in etchant and TNT arrays are fragile when they are attached to a substrate because of their amorphous crystallinity. In the physical etching method, the bottom layer of TNT arrays was removed by the plasma or ion milling process, which is not simple. However, the surface morphology and length of TNT arrays are not damaged in the process and they are very stable when they are attached to a substrate because TNT arrays have the ability to change crystallinity from the amorphous to the anatase phase. After UV irradiation using a silver source, ~30 nm Ag NPs were seen in the channels of freestanding TiO₂ nanotubes in high-angle annular dark-field (HAADF) images, as shown in Figure 2d. The length of the TiO₂ nanotubes was ~22 μm and the length of the scattering layer, which consisted of 400 nm TiO₂ NPs, was ~10 μm .

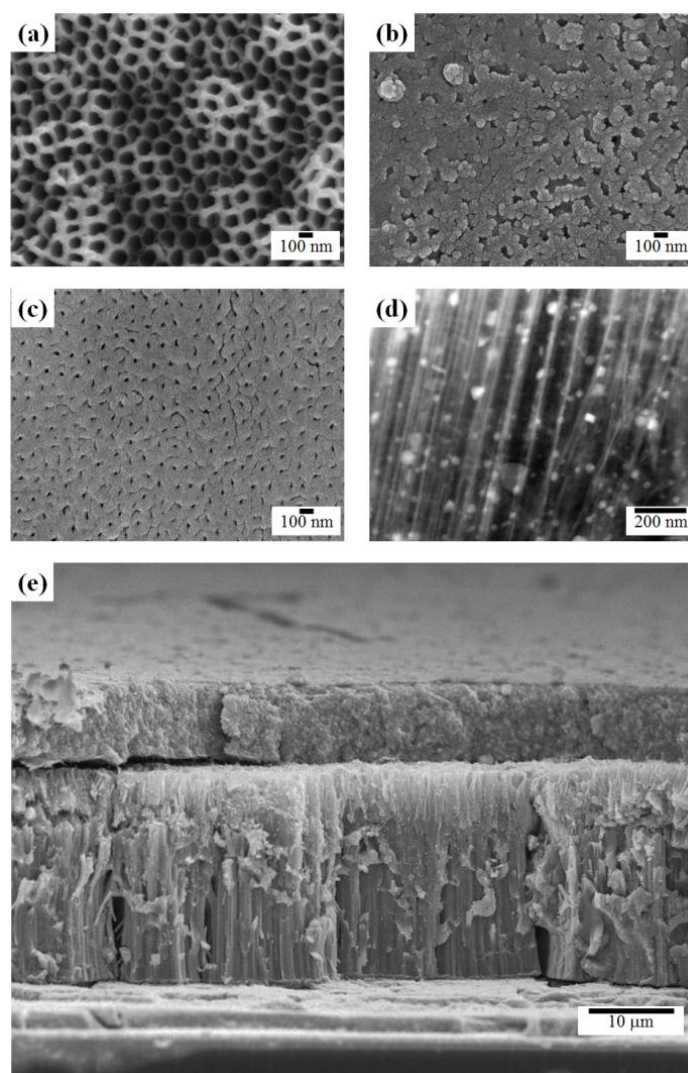


Figure 2. Field emission scanning electron microscope (FE-SEM) images of the (a) top, (b) bottom, and (c) bottom of post-ion milling freestanding TiO₂ nanotube arrays; (d) a high-angle annular dark-field (HAADF) image of Ag NPs in the channel of TiO₂ nanotube arrays; and (e) a side view of the active layer with freestanding TiO₂ nanotube arrays and a scattering layer.

The ultraviolet-visible (UV-vis) spectrum of Ag NPs in the channels of freestanding TiO₂ nanotubes is shown in Figure 3. A broad absorption peak centered at 402 nm was observed. The value is different from what it would be in the general solution phase, which is a 420 nm UV absorbance from 30 nm Ag NPs. This discrepancy may stem from different synthesis and measurement conditions used in this study [37–39]; the Ag NPs were synthesized by UV irradiation (at 254 nm) without adding a stabilizer and the Ag NPs were measured under dry conditions. The absorption band of Ag NPs is matched with the dye, *cis*-diisothiocyanato-bis(2,2'-bipyridyl-4,4'-dicarboxylato) ruthenium(II) bis(tetrabutylammonium) (N719) has two visible absorption bands, 390 nm and 531 nm, [40] that were affected by the plasmon band. Moreover, the shell of Ag NPs was prepared with TiCl₄ to prevent the trapping of electrons by Ag NPs and to enable better electron transport in the channel of the TiO₂ nanotube arrays.

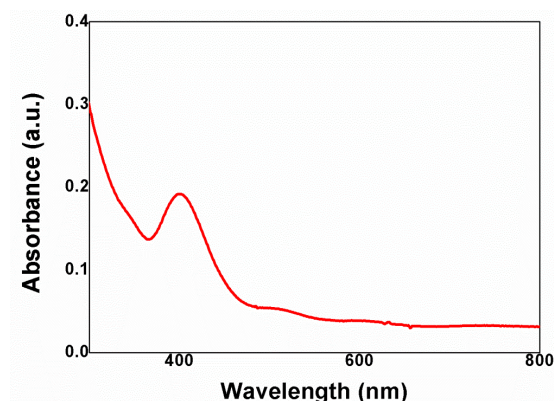


Figure 3. Ultraviolet-visible (UV-vis) spectrum of Ag NP-functionalized TiO₂ nanotubes.

2.3. DSSCs with Closed-Ended Freestanding TiO₂ Nanotube Arrays with Channels Containing Ag NPs and Large TiO₂ NPs

The photocurrent-voltage curves of DSSCs fabricated using closed-ended freestanding TiO₂ nanotube arrays measured under air mass 1.5 illumination (100 mW/cm²) are shown in Figure 4 and Table 1. Four types of closed-ended freestanding TiO₂ nanotube array-based DSSCs were fabricated in order to assess the effect of each component on the energy conversion efficiency: closed-ended freestanding TiO₂ nanotube array-based DSSCs without Ag or large TiO₂ NPs (a), with Ag NPs (b), with large TiO₂ NPs (c), and with both Ag NPs and large TiO₂ NPs (d). The open-circuit voltage (V_{oc}), short-circuit current (J_{sc}), fill factor (ff), and energy conversion efficiency (η) values are shown in Table 1. The energy conversion efficiency of DSSCs based on closed-ended freestanding TiO₂ nanotube arrays lacking NPs was 5.86%. The energy conversion efficiencies of DSSCs based on closed-ended freestanding TiO₂ nanotube arrays with Ag NPs, with large TiO₂ NPs, and with both Ag NPs and large TiO₂ NPs were 6.40%, 6.24%, and 6.71%, respectively. The introduction of Ag NPs increased the energy conversion efficiency significantly, by 9.21% (actual efficiency change, from 5.86% to 6.40%) compared to closed-ended freestanding TiO₂ nanotube array-based DSSCs without Ag and large TiO₂ NPs, because of increased light harvesting by the plasmonic effect. The introduction of large TiO₂ NPs also increased the energy conversion efficiency significantly, by 6.48% (actual efficiency, from 5.86% to 6.24%), owing to increased light harvesting by the scattering effect. Moreover, the introduction of both Ag NPs and large TiO₂ NPs increased the energy conversion efficiency significantly, by 14.50% (actual efficiency, from 5.86% to 6.71%), because of increased light harvesting resulting from both the plasmonic and scattering effects.

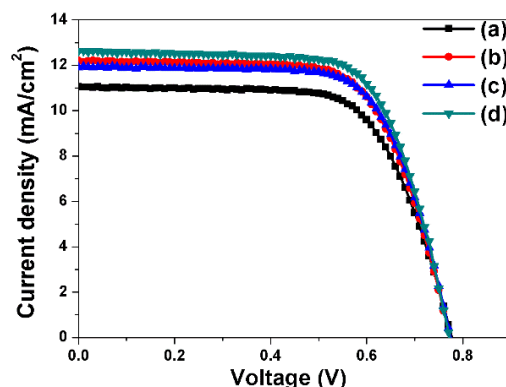


Figure 4. I–V curves of DSSC-based closed-ended freestanding TiO₂ nanotube arrays fabricated without NPs (a), with Ag NPs (b), with large TiO₂ NPs (c), and with Ag NPs and large TiO₂ NPs (d).

Table 1. Photovoltaic properties of dye-sensitized solar cells (DSSCs) based on closed-ended freestanding TiO₂ nanotube arrays.

DSSCs	J_{sc} (mA/cm ²)	V_{oc} (V)	ff	η (%)
(a) Closed-ended freestanding TiO ₂ nanotube arrays without any NPs	11.05	0.78	0.68	5.86
(b) Closed-ended freestanding TiO ₂ nanotube arrays with Ag NPs	12.22	0.77	0.68	6.40
(c) Closed-ended freestanding TiO ₂ nanotube arrays with large TiO ₂ NPs	11.90	0.76	0.69	6.24
(d) Closed-ended freestanding TiO ₂ nanotube arrays with Ag NPs and large TiO ₂ NPs	12.63	0.77	0.69	6.71

2.4. DSSCs with Open-Ended Freestanding TiO₂ Nanotube Arrays with Channels Containing Ag NPs and Large TiO₂ NPs

The photocurrent-voltage curves of DSSCs fabricated using open-ended freestanding TiO₂ nanotube arrays are shown in Figure 5 and Table 2; they are useful in assessing the effect of each component on the energy conversion efficiency. Four types of DSSCs based on open-ended freestanding TiO₂ nanotube arrays were fabricated: open-ended freestanding TiO₂ nanotube array-based DSSCs without Ag or large TiO₂ NPs (a), with Ag NPs (b), with large TiO₂ NPs (c), and with both Ag NPs and large TiO₂ NPs (d). The V_{oc} , J_{sc} , ff , and η values are summarized in Table 2. The energy conversion efficiency of DSSCs based on open-ended freestanding TiO₂ nanotube arrays lacking NPs was 6.12%. The energy conversion efficiencies of DSSCs based on open-ended freestanding TiO₂ nanotube arrays with Ag NPs, with large TiO₂ NPs, and with both Ag NPs and large TiO₂ NPs were 6.68%, 6.62%, and 7.05%, respectively. The introduction of Ag NPs, large TiO₂ NPs, and both increased the energy conversion efficiency by 9.15%, 8.17%, and 15.20%, respectively. Compared to closed-ended freestanding TiO₂ nanotube arrays, the energy conversion efficiency of DSSCs based on open-ended freestanding TiO₂ nanotube arrays was 5.07% (6.71%–7.05%) higher due to enhanced electron transport and electrolyte diffusion [33].

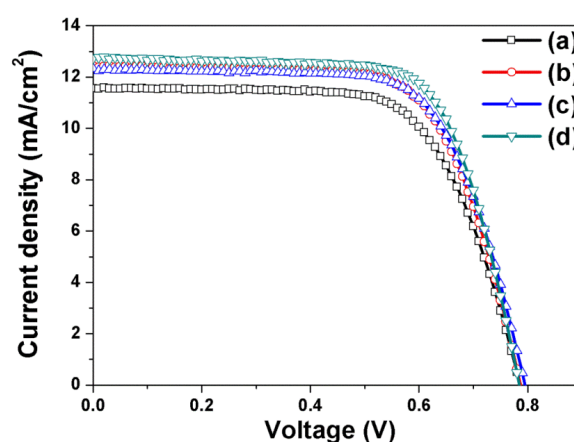
**Figure 5.** I–V curves of DSSCs based on open-ended freestanding TiO₂ nanotube arrays fabricated without NPs (a), with Ag NPs (b), with large TiO₂ NPs (c), and with both Ag NPs and large TiO₂ NPs (d).

Table 2. Photovoltaic properties of DSSCs based on open-ended freestanding TiO₂ nanotube arrays.

ADSSCs	J_{sc} (mA/cm ²)	V_{oc} (V)	ff	η (%)
(a) Open-ended freestanding TiO ₂ nanotube arrays without any NPs	11.56	0.79	0.67	6.12
(b) Open-ended freestanding TiO ₂ nanotube arrays with Ag NPs	12.45	0.79	0.68	6.68
(c) Open-ended freestanding TiO ₂ nanotube arrays with large TiO ₂ NPs	12.33	0.79	0.68	6.62
(d) Open-ended freestanding TiO ₂ nanotube arrays with Ag NPs and large TiO ₂ NPs	12.74	0.78	0.71	7.05

Although TiO₂ nanotube array-based DSSCs have great potential, as far as we know, the theoretical maximum improvement by Ag NPs or a TiO₂ scattering layer of TiO₂ nanotube-based DSSCs has not yet been reported. However, the open-ended TiO₂ nanotube-based devices exhibited an increase in one-sun efficiency from 5.3% to 9.1%, corresponding to a 70% increase, which is a much higher increase than we achieved [35]. We believe that there is ample room to improve efficiency by combining each parameter in an optimal condition based on theoretical studies.

3. Materials and Methods

3.1. Materials

Titanium plates (99.7% purity, 0.25 mm thickness, Alfa Aesar, Ward Hill, MA, USA), ammonium fluoride (NH₄F, Showa Chemical Industry Co., Beijing, China, 97.0%), ethylene glycol (Daejung Chemical, Siheung, Korea, 99%), hydrogen peroxide (H₂O₂, Daejung Chemical, Siheung, Korea, 30%), fluorine-doped tin oxide (FTO) glass (Pilkington, St. Helens, UK, TEC-A7), titanium diisopropoxide bis(acetylacetonate) solution (Aldrich, St. Louis, MS, USA, 75 wt % in isopropanol), n-butanol (Daejung Chemical, Siheung, Korea, 99%), TiO₂ paste (Ti-Nanoxide T/SP, Solaronix, Aubonne, Switzerland), scattering TiO₂ paste (18NR-AO, Dyesol, Queanbeyan, Australia), silver nitrate (AgNO₃, Aldrich, St. Louis, MS, USA, 99%), titanium chloride (TiCl₄, Aldrich, St. Louis, MS, USA, 0.09 M in 20% HCl), dye (cis-diisothiocyanato-bis(2,2'-bipyridyl-4,4'-dicarboxylato) ruthenium(II) bis(tetrabutylammonium) (N719, Solaronix, Aubonne, Switzerland), chloroplatinic acid hexahydrate (H₂PtCl₆ · 6H₂O, Aldrich, St. Louis, MS, USA), 1-butyl-3-methyl-imidazolium iodide (BMII, Aldrich, St. Louis, MS, USA, 99%), iodine (I₂, Aldrich, St. Louis, MS, USA, 99%), guanidium thiocyanate (GSCN, Aldrich, St. Louis, MS, USA, 99%), 4-tertbutylpyridine (TBP, Aldrich, St. Louis, MS, USA, 96%), acetonitrile (CH₃CN, Aldrich, St. Louis, MS, USA, 99.8%), and valeronitrile (CH₃(CH₂)₃CN, Aldrich, St. Louis, MS, USA, 99.5%) were obtained from commercial manufacturers.

3.2. Preparation of Freestanding TiO₂ Nanotube Arrays

TiO₂ nanotube arrays were prepared by anodization of thin Ti plates. Ti anodization was carried out in an electrolyte composed of 0.8 wt % NH₄F and 2 vol % H₂O in ethylene glycol at 25 °C and at a constant voltage of 60 V direct current (DC) for 2 h. After being anodized, the Ti plates were annealed at 500 °C for 1 h under ambient conditions to improve the crystallinity of the TiO₂ nanotube arrays, and then a secondary anodization was conducted at a constant voltage of 30 V DC for 10 min. The Ti plates were immersed in 10% H₂O₂ solution for several hours in order to detach the TiO₂ nanotube arrays from the Ti plates and produce freestanding TiO₂ nanotube arrays. The bottom of the TiO₂ nanotube arrays was removed by ion milling with Ar⁺ bombardment for several minutes in order to prepare open-ended freestanding TiO₂ nanotube arrays.

3.3. Preparation of Ag NPs in the Channel of Freestanding TiO₂ Nanotube Arrays

A TiO₂ blocking layer was formed on FTO glass by spin-coating with 5 wt % titanium di-isopropoxide bis(acetylacetonate) in butanol, followed by annealing at 500 °C for 1 h under ambient conditions to induce crystallinity. A TiO₂ paste was coated onto the TiO₂ blocking/FTO glass using the doctor blade method, and closed- and open-ended freestanding TiO₂ nanotube arrays were then introduced onto the TiO₂ paste. The substrate was annealed at 500 °C for 1 h under ambient conditions to enhance the adhesion between the TiO₂ NPs and freestanding TiO₂ nanotube arrays. The substrate was dipped in 0.3 mM AgNO₃ aqueous solution and exposed to 254 nm UV irradiation. The substrate was treated with 10 mM TiCl₄ solution at 50 °C for 30 min and then annealed at 500 °C for 1 h.

3.4. Fabrication of DSSCs with Freestanding TiO₂ Nanotube Arrays with Channels Containing Ag NPs

A substrate that consisted of freestanding TiO₂ nanotube arrays with channels containing Ag NPs was coated with ~400 nm TiO₂ NPs for scattering and then annealed at 500 °C for 1 h under ambient conditions to induce crystallinity and adhesion. The substrate was immersed in a dye solution at 50 °C for 8 h to function as a working electrode. The working electrode was sandwiched with a counter electrode, Pt on FTO glass, by a 60-μm-thick hot-melt sheet and filled with electrolyte solution composed of 0.7 M 1-butyl-3-methyl-imidazolium iodide (BMII), 0.03 M I₂, 0.1 M guanidium thiocyanate (GSCN), and 0.5 M 4-tertbutylpyridine (TBP) in a mixture of acetonitrile and valeronitrile (85:15, *v/v*).

3.5. Instruments

The morphology, thickness, size, and presence of Ag NPs in the channels of freestanding TiO₂ nanotube arrays were confirmed using a field emission scanning electron microscope (FE-SEM, JSM-6330F, JEOL Inc., Tokyo, Japan) and the high angular annular dark field (HAADF) technique with a scanning transmission electron microscope (TEM, JEM-2200FS, JEOL Inc., Tokyo, Japan). The current density–voltage (*J–V*) characteristics and the incident photon-to-current conversion efficiency (IPCE) of the DSSCs were measured using an electrometer (Keithley 2400, Keithley Instruments, Inc., Cleveland, OH, USA) under AM 1.5 illumination (100 mW/cm²) provided by a solar simulator (1 kW xenon with AM 1.5 filter, PEC-L01, Peccell Technologies, Inc., Yokohama, Kanagawa, Japan), or using a solar cell IPCE measurement System (K3100, McScience Inc., Suwon, Korea) with reference to the calibrated diode.

4. Conclusions

In this study, we compared the natural consequences of altering three parameters, the plasmonic effect, the scattering effect, and open- *vs.* closed-ended freestanding TiO₂ nanotubes, as a basic means of exploring improvements in efficiency. We demonstrated that the plasmonic and scattering effects enhanced the energy conversion efficiency of freestanding TiO₂ nanotube arrays in DSSCs. Ag NPs were added to the channels of TiO₂ nanotube arrays by UV irradiation to induce a plasmonic effect, and large TiO₂ NPs were introduced to TiO₂ nanotube arrays to induce a scattering effect. The energy conversion efficiency of DSSCs with both Ag NPs and large TiO₂ NPs was higher than that of DSSCs without Ag NPs owing to the plasmonic effect, and it was higher than that of DSSCs without large TiO₂ NPs owing to the scattering effect. Compared to closed-ended freestanding TiO₂ nanotube arrays, open-ended freestanding TiO₂ nanotube arrays [40] exhibited enhanced energy conversion efficiency. We demonstrate that Ag NPs, TiO₂ NPs, and open-ended freestanding TiO₂ nanotube arrays enhanced the energy conversion efficiency; furthermore, the combination of all components exhibited the highest energy conversion efficiency. Our research suggests that the energy conversion efficiency of DSSCs is improved by both the plasmonic and scattering effects. This knowledge has applications in organic solar cells, hybrid solar cells, and perovskite solar cells.

Author Contributions: Won-Yeop Rho, Myeung-Hwan Chun, Jung Sang Suh and Bong-Hyun Jun designed the research and wrote the manuscript. Won-Yeop Rho, Ho-Sub Kim and Hyung-Mo Kim carried out experiments. All authors discussed the results and commented on the manuscript. Won-Yeop Rho and Bong-Hyun Jun guided all aspects of the work.

Conflicts of Interest: The authors declare no conflict of interest.

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