

Article

# Synthesis and Optimization of Multiwalled Carbon Nanotubes–Ferrihydrite Hybrid Composite

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**Abstract:** Carbon nanotubes (CNT) are a family of carbon nanomaterials that have uses in many technological and medical applications due to their unique properties. However, compared to other nanomaterials, CNT have a significantly lower specific surface areas (SSA), which is a critical limitation for applications. To overcome this limitation, here, we report a new protocol to synthesize a hybrid material composed of varying ratios of multiwalled carbon nanotubes (MWCNT) and ferrihydrite (FHY). Furthermore, through a series of physical and electrochemical characterization tests, we determined that 36% FHY and 64% MWCNT is the optimum ratio for a composite that maximizes both SSA and specific capacitance. The calculated SSA of the composite was  $190 \text{ m}^2 \cdot \text{g}^{-1}$ , 2.9 times higher than that of MWCNT alone. Moreover, the composite retained valuable electrochemical properties of CNT with an estimated specific capacitance of  $100 \text{ F} \cdot \text{g}^{-1}$ . This composite is a promising multifunctional nanomaterial for environmental and technological applications requiring electrochemical reactivity and high specific areas such as environmental biosensors, and capacitive deionization for wastewater remediation, and water softening.

**Keywords:** electrochemistry; surface modification; environmental nanotechnology; characterization; nanocomposites; material chemistry



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## 1. Introduction

In 1991, carbon nanotubes (CNT) was characterized by Iijima [1] and have since received increased attention in both research and development due to their remarkable electrical conductivity, wide thermal operating range, chemical and physical stability, and high aspect ratios [2–4]. Unlike other conventional carbon materials, CNT have a 3D structure of cylindrically rolled-up chicken wire tubes made from covalently bonded carbon atoms resulting in the two main types of CNT: (1) single-walled CNT (SWCNT) consisting of a single layer of cylindrical graphene sheet [5,6] and (2) multiwalled carbon nanotubes (MWCNT) consisting of multilayered graphene sheets [1,6].

Due to their desirable high aspect ratios and electrochemical performance, CNT is good candidate for many applications, including as potential adsorbents for contaminants in water and supercapacitor electrode materials [6–10]. The reported specific surface area (SSA) for MWCNT range from  $170$  to  $280 \text{ m}^2 \cdot \text{g}^{-1}$  [11], whereas values for activated carbon nanomaterials, for instance, are higher than  $500 \text{ m}^2 \cdot \text{g}^{-1}$  [12]. MWCNT's overall lower SSA limits their popularity for commercial applications. To overcome this obstacle, some researchers have developed CNT-based hybrid materials that showed a significant improvement in their surface area by including materials such as mesoporous carbon, activated carbon, nanofibers, nanoporous anodic alumina, and ferrihydrite (FHY) [13–18].

FHY is a naturally occurring nanocrystalline iron oxyhydroxide, with reported SSA up to  $650 \text{ m}^2 \cdot \text{g}^{-1}$  [19]. It plays a significant role in trapping contaminants from groundwater and moving them across terrestrial environments [20,21]. Several studies have combined FHY with MWCNT [17,18] to improve the adsorption capacity of MWCNT in adsorbing

aqueous contaminants. However, these studies only considered a single point of composition ratio between MWCNT and FHY and assessed changes of only the physical properties, i.e., crystal structure, morphology, and SSA, of the hybrid mixture [17,18]. Moreover, no other studies investigated the electrochemical performance of an FHY/MWCNT composite. The novelty of this work is that we expand on past studies by synthesizing composites of FHY and MWCNT at five different composition ratios and present a comprehensive characterization and optimization of their physical and electrochemical properties. To our knowledge, this is the first study to comprehensively characterize the promising FHY/MWCNT material. The detailed characterization encompassed determination of the SSA by a Brunauer–Emmett–Teller (BET) analysis, morphologies by transmission electron microscopy (TEM), crystallinity by X-ray powder diffraction (XRD), and electrochemical properties by cyclic voltammetry (CV). Additionally, the composites exhibited physico-chemical properties, e.g., density and dispersibility, that proved problematic in preparation procedures traditionally used for the characterization methods we utilized. Thus, this study also reports on newly developed protocols for synthesizing and characterizing the physical, chemical, and electrochemical properties of these composite materials. These procedures may be adopted in future studies that seek to develop and investigate these composites or others containing metal oxides/hydroxides.

## 2. Materials and Methods

### 2.1. Materials and Preparation

Prior to the experiment, all equipment and glassware were treated with ethanol and 10% nitric acid solution to reduce organic and metal contaminations, respectively [22]. MWCNT (> 97% purity) were purchased from Shenzhen Nanotech Port Co. Ltd. (Shenzhen, China) and manufactured by catalytic decomposition of methane. According to manufacturer MSDS, their outer diameter ranged between 40 and 60 nm and lengths between 5 and 15  $\mu\text{m}$ .

MWCNT were refluxed in 2.6 M  $\text{HNO}_3$  solution at 140  $^\circ\text{C}$  for 24 h and then washed repeatedly under vacuum filtration with 18.2 M $\Omega\cdot\text{cm}$  deionized  $\text{H}_2\text{O}$  until the effluent pH reached  $7.0 \pm 0.2$ . The MWCNT was dried in an oven at 100  $^\circ\text{C}$  for 24 h. The acid refluxing step was intended to remove any residual catalyst or metal impurities from the manufacturing processes and functionalize the surface of the MWCNT with carboxyl groups. This functionalization was necessary to enhance the dispersion, surface area, and electrical properties of the MWCNT [23–25].

### 2.2. FHY/MWCNT Composite Synthesis

The composite materials were synthesized at 5 wt% ratios of FHY/MWCNT 100/0, 75/25, 50/50, 25/75, and 0/100 as follows. Appropriate masses of MWCNT and iron nitrate ( $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ , ACS grade, ACROS Organics) were mixed in glass beakers, then added to 200 mL of 18.2 M $\Omega\cdot\text{cm}$   $\text{H}_2\text{O}$ , magnetically stirred for 10 min, and sonicated for another 30 min for even dispersion. These well-dispersed suspensions were treated according to a modified Schwertmann and Cornell's 2-line FHY synthesis protocol [26], which calls for titrating with 1 M potassium hydroxide solution (KOH, Laboratory Grade, Carolina) to pH  $7.5 \pm 0.2$  to induce FHY precipitation. The precipitates were washed repeatedly with 18.2 M $\Omega\cdot\text{cm}$   $\text{H}_2\text{O}$  using vacuum filtration until the conductivity of effluent solutions reached a value below 20  $\mu\text{S}\cdot\text{cm}^{-1}$ . On the average, analyzed samples showed that most of the  $\text{Fe}^{3+}_{(\text{aq})}$  (mass balance, >92%) was driven into a solid phase and precipitated onto the surface of MWCNT. The synthesized composites were dried first under a vacuum desiccator for 24 h and then under a fume hood for an additional 24 h to ensure complete water evaporation. The dry composites were stored into sealed containers that were placed in a freezer at  $-18\text{ }^\circ\text{C}$  pending for analysis.

### 2.3. X-ray Powder Diffraction (XRD) Analysis

Samples were analyzed on a XPERT PRO MPD X-ray diffractometer (Malvern PANalytical, Malvern, UK), operating at 45 kV and 40 mA, from 5° to 80° 2 $\theta$ , using a Cu anode and a graphite crystal monochromator to eliminate Fe fluorescence.

### 2.4. Transmission Electron Microscope (TEM) Analysis

TEM was used to study the surface morphologies, structure orientations, topography, and uniformity of the composite. Approximately a solution of 0.5 mg/mL of samples in ethanol was dispersed using sonication under >20 kHz for a few minutes. These suspensions were then dropped on a LC300-Cu-150 TEM grid, with 150 nm openings lacey C support (Electron Microscopy Sciences, Hatfield, PA, USA) for successive TEM analysis. Samples were analyzed on a JEOL-2100 LaB6 TEM (JEOL, Peabody, MA, USA), with a point resolution of 0.194 nm, operated at 200 keV.

The TEM is equipped with a JEOL EX-230 Silicon Drift Detector (JEOL, Peabody, MA, USA) with a 60 mm<sup>2</sup> window of acquisition for energy dispersive X-ray analysis (EDS). Scanning transmission electron microscopy (STEM) digital scanning images were acquired while collecting EDS maps with a JEOL EM-245111SIOD (JEOL, Peabody, MA, USA) bright-field STEM detector that has a resolution of 1 nm.

### 2.5. Brunauer–Emmett–Teller (BET) Surface Area Analysis

The specific surface areas of the dried composites were evaluated by conducting a 7-point BET N<sub>2</sub> adsorption isotherms at 77 K ET using a Quantachrome Autosorb-1 (Quantachrome Instruments, Boynton Beach, FL, USA). Before the measurements, the samples were degassed overnight at 80 °C. A low degassing temperature was selected to prevent the removal of the structural water of FHY and hence its transformation. The sample mass utilized for the analysis varied from 0.08 to 0.31 g, depending on the density of each composite ratio.

### 2.6. Cyclic Voltammetry (CV) Analysis

#### 2.6.1. Thin-Film Deposition for Working Electrodes (WE)

CV was used to analyze the specific capacitance of each composite at various scan rates. In order to perform the CV measurements, it was important to develop an effective method of depositing the composites on metal plate substrates. MWCNT have been a popular material for supercapacitors, and there are several standard methods for MWCNT thin-film deposition [27–31]. However, the FHY/MWCNT composites had new electrochemical properties compared to either CNTs or FHY alone. Attempts at conventional thin-film deposition techniques such as chemical vapor deposition, electrophoretic deposition, and doctor blade technique were unsuccessful. To solve this problem, a 5 wt% Nafion ionomer solution (C<sub>7</sub>HF<sub>13</sub>O<sub>5</sub>S·C<sub>2</sub>F<sub>4</sub>, Ion Power) was used as a binder to deposit the composites on stainless steel substrate plates. Nafion has been used in previous electrochemical studies as a binder that does not interfere with the signal of target analytes [32–34]. Working electrodes (WE) were made from stainless steel (ASTM A240, 16 Ga) with dimensions of 1.5 cm × 0.19 cm × 1 cm. These electrodes were cleaned by acetone prior to the CV analysis.

Suspensions combining each composite and Nafion at a density of 2 mg·mL<sup>−1</sup> were prepared and sonicated for at least 30 min. Then, 50  $\mu$ L of these mixtures was drop-casted on each side of the electrodes. The electrodes were placed on a rack in a sealed chamber and suspended over an ethanol bath to avoid cracking due to rapid drying. The elevated ethanol vapor pressure in the chamber slowed the evaporation rate sufficiently to allow the composite to deposit evenly on the surface of the electrodes. These electrodes were then air dried for at least 15 more min before the CV experiments to ensure complete evaporation. Three WE samples were made at each composite ratio. The deposited amount of material on these electrodes were recorded for later mass-based calculations; the deposited mass was kept under 5 mg.

### 2.6.2. CV Setup

A standardized potentiostat (Reference 600, Gamry instruments, PA, USA) was used for the CV analysis in a standard three-electrode system. The WEs were the stainless-steel plates with composite material deposited as described in Section 2.6.1. The counter electrode (CE) was a platinum plate (Gesswein, CT, USA) and the reference electrode (RE) Ag/AgCl in 3 M saturated NaCl solution (BASi, IN, USA)

The potentials of WE were measured against the RE with negatively charged CE. The electrolyte in these experiments was 0.5 M Na<sub>2</sub>SO<sub>4</sub> solution. Unlike other organic electrolytes, Na<sub>2</sub>SO<sub>4</sub> solution has a lower equivalent series resistance and pore size requirement [35]. Measurements were made within a potential window of 1 V with an upper and lower limit of 0.5 and −0.5 V, respectively. This potential window was limited to the 1 V range because water could breakdown at 1.23 V [36]. Each composite was run three times and scanned at the following scan rates of 100, 70, 50, 20, and 2 mV·s<sup>−1</sup>. The following equation was used to determine the specific capacitance (C<sub>s</sub>, in F·g<sup>−1</sup>) of samples at various sweep rates [35,37].

$$C_s = \frac{\int_{V_a}^{V_b} i(V) dV}{2(V_b - V_a)vm}, \quad (1)$$

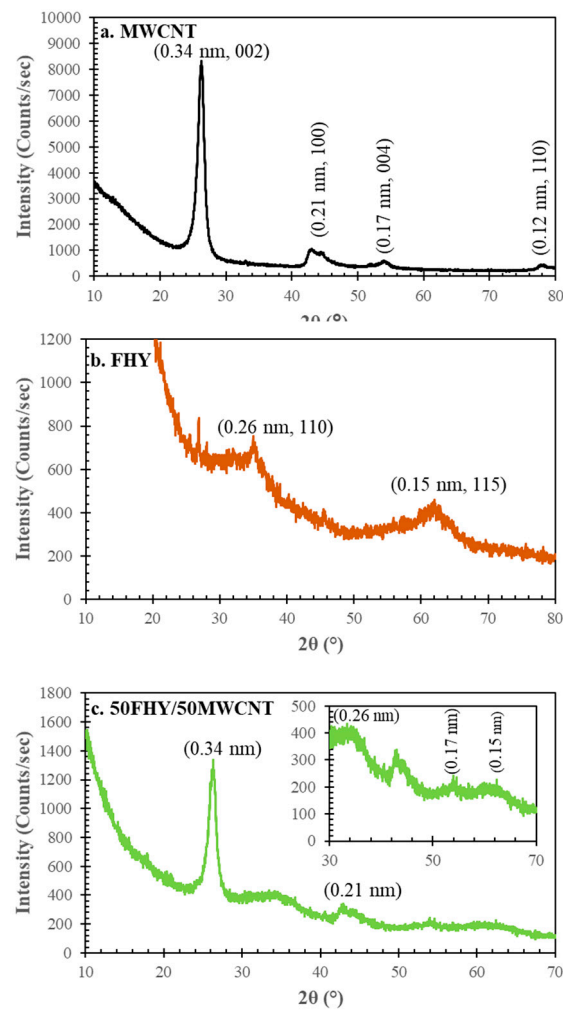
where  $\int_{V_a}^{V_b} i(V) dV$  was the integrated area of the CV plots;  $v$  (in V·s<sup>−1</sup>) was the scan rate;  $m$  (in g) was the mass of loaded material on the WE; and  $V_b$  and  $V_a$  (in V) were the upper and lower voltage limit of the scan, respectively.

## 3. Results and Discussion

### 3.1. XRD Patterns of the FHY/MWCNT Composite

Figure 1 shows representative X-ray diffractograms for the 0/100, 50/50, and 100/0 FHY/MWCNT materials. MWCNT presented a high-intensity diffraction peak around 26° ( $d = 0.34$  nm), and three lower intensity diffraction peaks around 44° ( $d = 0.21$  nm), 53° ( $d = 0.17$  nm), and 78° ( $d = 0.12$  nm) (Figure 1a). These peaks correspond respectively to the (002), (100), (004), and (110) diffraction patterns of typical graphite, which indicates that the acid-treated MWCNT in this study were well graphitized [38]. All these peaks, except the (110), were present in the pattern of the 50/50 FHY/MWCNT composite (Figure 1c). Based on the similarity of our XRD patterns with those from Atchudan et al. (2015), we conclude equally that neither MWCNT nor FHY/MWCNT composites contain significant carbonaceous or metal impurities. We conclude that the quality of the starting materials is not compromised during the synthesis of the composite material. In Figure 1b, the pattern for pure FHY showed two main diffraction peaks around 34° ( $d = 0.26$  nm) and 61° ( $d = 0.15$  nm) corresponding, respectively to the (110) and (115) diffraction patterns of a 2-line FHY [39]. These peaks were also evident in Figure 1c of the FHY/MWCNT composite revealing FHY was successfully synthesized in the composite.

The XRD data indicate that the FHY/MWCNT composite inherited the characteristics XRD patterns of both FHY and MWCNT. Hence, the synthesis procedure we developed for the composite materials did not interfere with the formation of FHY or the integrity of the MWCNT. Moreover, the crystallinity signature of each composite samples is qualitatively representative of each of its end-member.



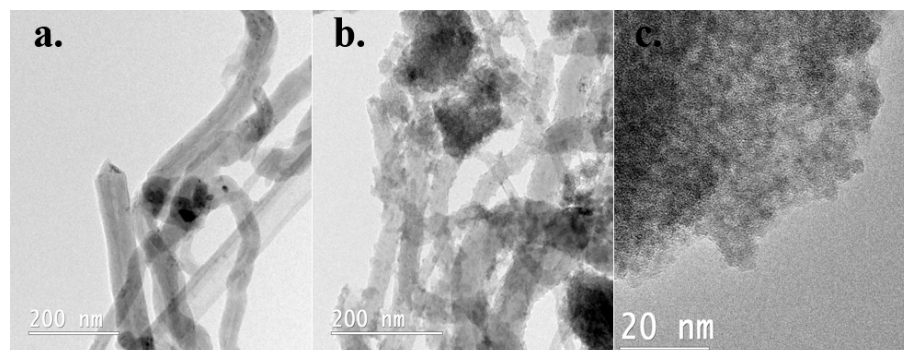
**Figure 1.** XRD patterns of (a) acid-treated multiwalled carbon nanotubes (MWCNT), (b) synthesized ferrihydrite (FHY) (blue), and (c) 50 FHY/50 MWCNT composite; the inset in (c) showed smaller intensity XRD peaks at smaller scale for better illustration. All XRD peaks in these figures were labeled with their corresponding  $2\theta$  angles and d-spacings in nm, and their characterized XRD patterns from previous studies in (a) and (b).

### 3.2. TEM Results

Figure 2a shows the morphology of acid-treated MWCNT, which consisted of regular tubular structures of outer diameter in the range of 30–50 nm and length reported by the manufacturer to be 5–15  $\mu\text{m}$ . Figure 2c reveals the morphology of synthesized FHY particles, i.e., composite sample 100/0. FHY particles were nanocrystalline with a particle size estimated roughly at 3 nm, similar to two-line FHY reported in earlier studies [40–42]. The 50 FHY/50 MWCNT samples (Figure 2b) consisted of MWCNT covered with an agglomerated layer of FHY particles coating the surface of the tubes. While most of the tubes surface appears to be covered by FHY, some regions had higher density of FHY deposition as evident from the higher contrast regions. This morphology confirmed the presence of a two-phase solid made of FHY and MWCNT. Based on the distribution of FHY on the MWCNT seen in Figure 2b, we conclude that there is a strong bonding between these two materials in the composite materials since the FHY remained intact on the MWCNT despite the extensive sonication administered prior to TEM analysis. During the  $\text{HNO}_3$  refluxing step, the surface of MWCNT was functionalized with carboxylic groups with pKa values of 1.8–2.4 [43]. At the pH values of 6–7.0, reached during the synthesis and subsequent washing, we expect the carboxylic groups to fully deprotonate and the MWCNT to carry an overall negative charge. The point of zero charge ( $\text{pH}_{\text{pzc}}$ ) of FHY is



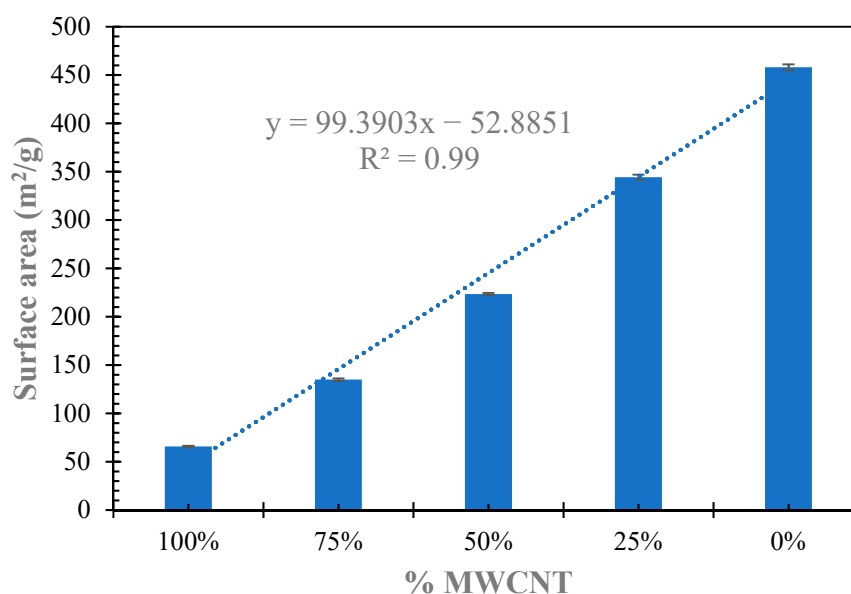
reported between 7.5 and 8.2 [44–46], and thus in our experiment, the synthesized FHY nanoparticles carry a positive charge. We postulate that ionic bonds between the carboxyl groups on the MWCNT and the hydroxyl groups on FHY underlie the attachment of FHY and MWCNT in the hybrid composites.



**Figure 2.** TEM images of (a) pure acid-treated MWCNT, (b) 50 FHY/50 MWCNT composite, and (c) pure synthesized FHY.

### 3.3. Surface Area (BET) and Specific Capacitance (CV) Analysis of the FHY/MWCNT Composites

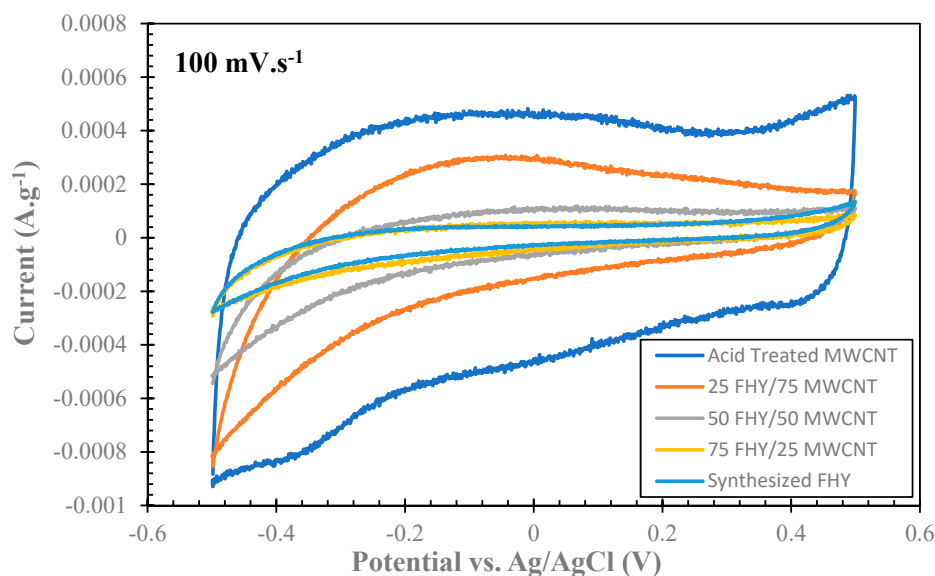
Figure 3 shows a strong correlation between the measured SSA and the % FHY in the composite material ( $R^2 = 0.99$ ). The SSA increased with increasing % FHY in the composite samples. Since FHY had the highest SSA of  $458.02 \text{ m}^2 \cdot \text{g}^{-1}$ , which was almost 7 times higher than that measured for MWCNT ( $65.76 \text{ m}^2 \cdot \text{g}^{-1}$ ), the observed trend supports the viability of incorporating FHY with MWCNT to increase SSA. On average, the surface area would increase by approximately 5.97% with every percent increase in % FHY in the composite ratio. Additionally, the SSA value obtained for untreated MWCNT ( $54.23 \text{ m}^2 \cdot \text{g}^{-1}$ ) was 1.2 times lower than that of acid-treated MWCNT, which indicated that refluxing MWCNT with  $\text{HNO}_3$  can also be used to increase the surface area of MWCNT material.



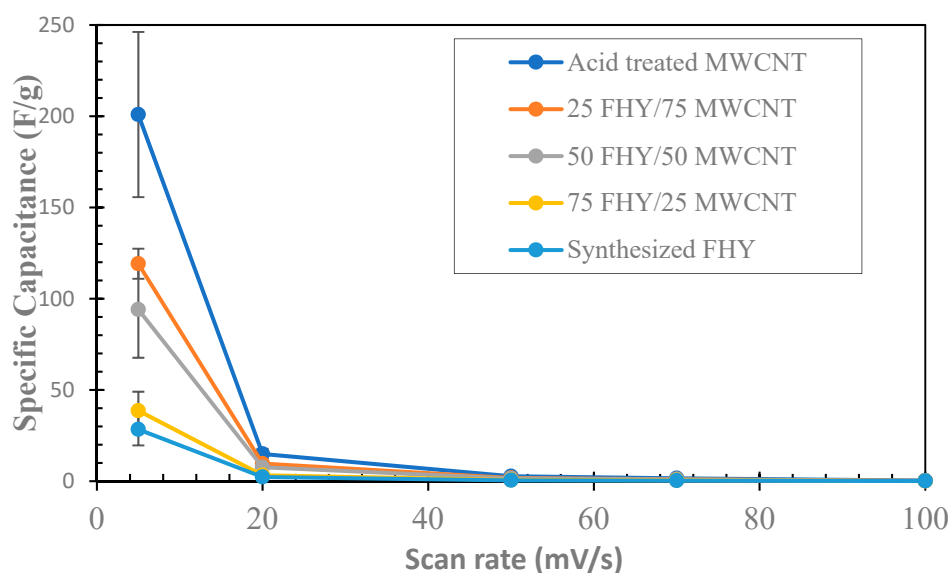
**Figure 3.** Specific surface area data for all composite and end-member samples. Trendline and equation showing the relationship between the % MWCNT in the composite material and its measured BET specific surface area. Error bars represent the standard deviation of triplicate samples.

Figure 4 shows the CV plots of all samples ran at a scan rate of  $100 \text{ mV} \cdot \text{s}^{-1}$ . The current density increased with increasing % MWCNT in all composite samples, indicating more electrolyte ions in the electrical double layer at the surface of the composite electrode with

higher % MWCNT [35,37]. Moreover, the CV plot of the MWCNT had a near rectangular shape indicating the characteristic of an effective capacitor [35,37]. The plots for the hybrid composite materials deviated further from the ideal rectangle shape with increases in % FHY suggesting a lower electrochemical performance in these samples. The negative impact of increasing % FHY on the electrochemical performance is also observed at all other scan rates tested (Figure 5).



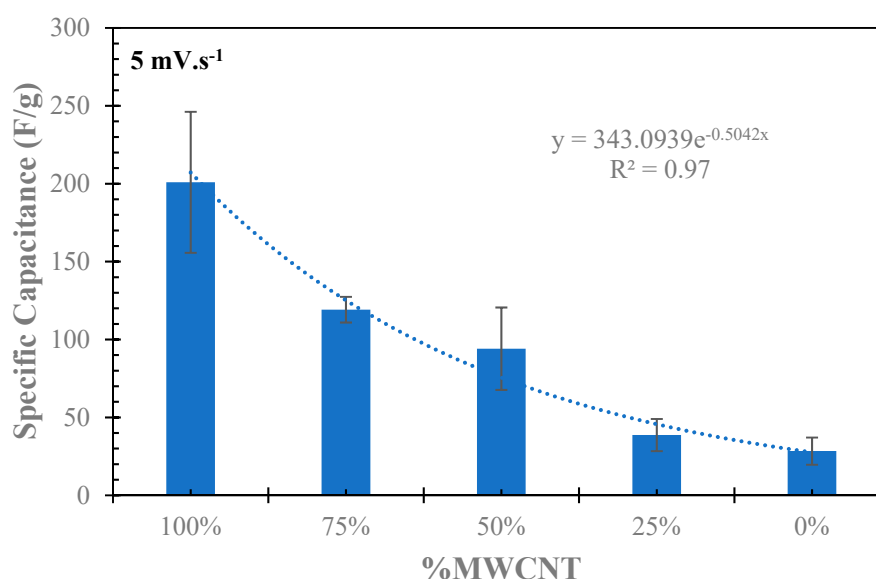
**Figure 4.** Cyclic voltammetry (CV) plot of all composite samples and controls at  $100 \text{ mV}\cdot\text{s}^{-1}$  in  $0.5 \text{ mol}\cdot\text{L}^{-1} \text{ Na}_2\text{SO}_4$  electrolyte solution.



**Figure 5.** Specific capacitance of all composite samples and controls at various scan rate. Error bars reveal the standard deviation obtained from triplicate electrodes measurements.

Figure 6 shows the highest observed specific capacitance of these samples at  $5 \text{ mV}\cdot\text{s}^{-1}$  for easier comparison. It is noteworthy to mention that the precision of 50 FHY/50 MWCNT may have been compromised by the loss of one sample in the three replications during the analysis. Instead of excluding the sample, we chose to include it to highlight the trend in the analysis. Obtaining CV data for these hybrid composites proved extremely challenging due to the difficulties in achieving evenly coated electrodes with no fractures despite the

use of a delayed-evaporation chamber. This challenge highlights the need to further improve the electrochemical characterization methods as this or other composite materials are synthesized and primed for future applications. As expected, the samples with higher % MWCNT exhibit higher specific capacitance indicating better electrochemical characteristics. At  $5 \text{ mV} \cdot \text{s}^{-1}$ , the calculated specific capacitance of MWCNT was  $200.91 \text{ F} \cdot \text{g}^{-1}$ , almost 7.1 times higher than that of the FHY alone ( $28.35 \text{ F} \cdot \text{g}^{-1}$ ). Furthermore, the effect of increasing MWCNT content on specific capacitance of the composite was close to exponential growth, with  $R^2 = 0.97$ . On average, the specific capacitance would increase 2.75% for every increased percent of MWCNT in the composite ratio.



**Figure 6.** Specific capacitance data of all composite samples and controls at the scan rate of  $5 \text{ mV} \cdot \text{s}^{-1}$ . Trendline and equation showing the relationship between composite ratio and specific capacitance were included.

### 3.4. Optimal FHY/MWCNT Composite Ratio Prediction

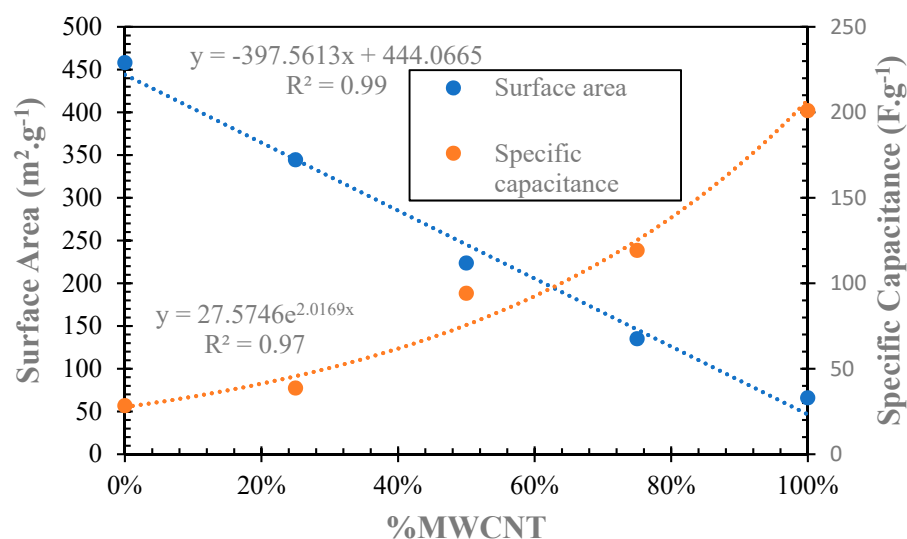
Taken cumulatively, our results indicate that both MWCNT and FHY impart desirable and undesirable properties on the composite materials. MWCNT enhance the specific capacitance but reduce the specific surface area, while the opposite effects are true for FHY. Therefore, to determine the optimum ratio for a composite material with the highest SSA and specific capacitance, we plotted SSA and specific capacitance simultaneously in search of the intersection point, which represent the optimal composition (Figure 7).

The relationship between SSA and specific capacitance with % MWCNT in the composite materials could be explained by the nature of the properties measured in the composite samples. BET  $\text{N}_2$  adsorption responds to changes in the surface of the composite materials. Therefore, if the materials are synthesized carefully and the target ratio of FHY to MWCNT in the composite is achieved, as it is in our study, the measured SSA will increase proportionally and linearly with the mass of FHY. In contrast, the pathways of ion movement during electrochemical measurements are potentially more complicated. These pathways determine the current density, and hence directly affect the calculated specific capacitance of the materials. As mentioned, based on TEM data, FHY coated most of the of MWCNT surface in the composite samples, but tended to be aggregated together in samples with higher % FHY. These FHY aggregates could have prevented the transport of solvent ions and reduced the observed specific capacitance of those materials.

The intersection between the two curves occurred at about 36% FHY and 64% MWCNT. This optimal composite ratio yields a calculated surface area of  $190 \text{ m}^2 \cdot \text{g}^{-1}$  and estimated specific capacitance of  $100 \text{ F} \cdot \text{g}^{-1}$  at  $5 \text{ mV} \cdot \text{s}^{-1}$ . In comparison, the SSA of this composite



was 2.9 times higher than that of MWCNT alone and about 3.5 times higher in specific capacitance at  $5 \text{ mV}\cdot\text{s}^{-1}$  than pure FHY.



**Figure 7.** The combination plot showed the fluctuation of both specific capacitance and specific surface area with different ratios of MWCNT and FHY in the composite material.

#### 4. Conclusions, Technical Implication, and Future Directions

The main goal of this study was developing a synthesis procedure to create a novel hybrid material that could address deficiencies of traditional MWCNT's SSA when compared to conventional carbon nanomaterials. Our results demonstrate that the SSA of MWCNT could be increased by doping it with FHY. Yet, to preserve a desirable electrochemical reactivity, we estimate an optimal composite ratio of 36% FHY and 64% MWCNT. This ratio will yield a hybrid material whose SSA is 2.9 times higher and specific capacitance 2.0 times lower than those of MWCNT alone.

It is noteworthy to mention again that this new material displayed unique physico-chemical properties (density and dispersibility) that could not be analyzed by traditional characterization methods especially for electrochemical (CV) analysis. Based on available resources and knowledge, the authors had tried their best to analyze the electrochemical aspects of this material. We believe future studies should be developed to perform more complex electrochemical analysis for this material and reveal more insights about this promising nanomaterial. For example, future studies could be directed at obtaining an even coating/adsorption of FHY onto MWCNT surfaces to improve on the specific capacitance and analysis of the hybrid material.

Moreover, this study stopped at synthesis and present convenient yet comprehensive and wholistic characterization of the FHY/MWCNT composite. Hence, future real application studies should be developed further to emphasize the novelty of this material. In fact, our team is developing different sorption studies of this nanomaterial with various environmental contaminants as our next steps. We also are going to investigate the biofouling merits of this material for environmental applications. Additionally, the newly synthesized material will also have direct uses in applications that already utilize CNT and other metal oxides, such as capacitive deionization technologies as applied to water treatment.

**Author Contributions:** Conceptualization, T.T. and D.M.D.; methodology, T.T. and D.M.D.; software, T.T.; validation, T.T., D.M.D., and N.K.; formal analysis, T.T.; investigation, T.T.; resources, D.M.D. and N.K.; data curation, T.T.; writing—original draft preparation, T.T.; writing—review and editing, T.T., D.M.D., and N.K.; visualization, T.T.; supervision, D.M.D. and N.K.; project administration, T.T. and D.M.D.; funding acquisition, D.M.D. and N.K. All authors have read and agreed to the published version of the manuscript.

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